# Short Communication Hollow Sphere c-In<sub>2</sub>O<sub>3</sub> for Application in Lithium Ion Battery

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The hollow sphere c-In<sub>2</sub>O<sub>3</sub> were synthesized via hydrothermal reaction. The crystal structure of the sample was collected and analyzed through X-ray diffractometry (XRD). The surface morphology and particle size of the sample were observed by scanning electron microscope (SEM). The electrochemical property of the hollow sphere c-In<sub>2</sub>O<sub>3</sub> as cathode for Li-ion Batteries were studied. Hollow Sphere c-In<sub>2</sub>O<sub>3</sub> shows a discharge capacity was 1162 mAh·g<sup>-1</sup> at the current density of 30 mA·g<sup>-1</sup> in a voltage of 0.2-1.4V, and the reversible capacity is around 100 mAh·g<sup>-1</sup> after 40 cycles.

**Keywords:** c-In<sub>2</sub>O<sub>3</sub>; hydrothermal; hollow sphere; metal oxide; lithium ion battery

## **1. INTRODUCTION**

In recent years, growing demands in various electronic devices, digital communication, hybrid electric vehicles, other related devices ,and high power source have been investigated[1]. As new potential anode of materials rechargeable lithium batteries, metal oxides been applicated to the area of the energy storage [2,3]. Due to unique physical, chemical properties [4], and higher theoretical reversible capacity (500~1000 mAh·g<sup>-1</sup>), metal oxides than that of the current commercial graphite (~370mAh·g<sup>-1</sup>)[5-8].

Hollow sphere of metal oxides have been attented for several years, because of their unique properties of specific structural, nanometer sizes, good permeation, low density, and high specific surface area [7]. Many methods have been developed to fabricate hollow sphere of metal oxides, for instance, pyrohydrolytic [9], hydrothermal method [10-15], thermal evaporation [16], chemical vapor deposition [17], pulsed laser deposition [2,18] and electrochemical deposition [18].

Indium oxide  $(In_2O_3)$  as promising cathode material has been attracted attention, owning to its high theoretical capacities, electrical conductivity and applications extensively [19-20]. In<sub>2</sub>O<sub>3</sub> is very important n-type semiconductor binary oxide with direct band gap of 3.55-3.75eV and indirect band gap around 2.8 eV, and has been widely applied in microelectronic areas including gas sensors, transparent conductors, solar cells, ultrasensitive toxic gas detectors, and liquid crystal display devices [9,10,21,22]. In<sub>2</sub>O<sub>3</sub> has three different crystal structures: cubic bixbyite-type (c-In2O3), hexagonal corundunm-type (h-In<sub>2</sub>O<sub>3</sub>) and orthorhombic Rh<sub>2</sub>O<sub>3</sub> (II)-type[23]. c-In<sub>2</sub>O<sub>3</sub> is stable under ambient conditions and the last structure is stable only under high pressures and temperatures [3,4,11].

In this work, we demonstrate a facile hydrothermal process for controllable synthesization of hollow sphere c-In2O3. The crystal structure and morphology of samples are observed through X-ray diffraction (XRD) and scanning electron micrographs (SEM). Moreover, the electrochemical properties of the hollow sphere c-In2O3 have been investigated as cathode materials for Li-ion Batteries. The results show that the as-prepared hollow sphere c-In2O3 exhibit high discharge capacity and coulombic efficiency.

### 2. EXPERIMENTAL METHODS

The hollow sphere c- $In_2O_3$  sample was synthetized by hydrothermal process: 0.381g  $In(NO)_3$ ·4.5H<sub>2</sub>O and 8g sucrose were dissolved in 30 mL distilled water. 1 mL of dimethylformamide was added into the former liquor under vigorous stirring for 1h, at room temperature. The resulting mixture was transferred into a Teflon-lined stainless steel autoclave of 50 mL capacity. The autoclave was then sealed and maintained at 80°C for 20h in an oven followed by cooling the autoclave down to room temperature naturally. After being cooled, the prepared white precipitate was collected by centrifugation and washed with distilled water and the absolute ethanol. The washing cycle was repeated several times, followed by drying at 80°C overnight. The product was white  $In(OH)_3$  sample. Finally, the yellow hollow sphere c- $In_2O_3$  sample was obtained from the white  $In(OH)_3$  sample via calcination at 550°C for 2h with eating rate of 2°C·min<sup>-1</sup>, and cooling rate of 1°C·min<sup>-1</sup>, in a muffle furnace.

The crystalline phase of  $In_2O_3$  was identified by an X-ray diffractometer (Bruker D8 Advance) using Cu K $\alpha$  radiation ( $\lambda$ =0.154184 nm) with 40 kV of voltage and 30 mA of current. XRD data was recorded at 2 $\theta$  with a step size of 0.03° in the range from 10° to 90°. The particle size and morphology of hollow sphere c-In<sub>2</sub>O<sub>3</sub> powders were observed with a scanning electron microscope.

The electrochemical performance of the hollow sphere  $c-In_2O_3$  powders as cathode was evaluated using a CR2016 coin-type cell with a lithium metal anode. The CR2016 coin-type cell was fabricated with the hollow sphere  $c-In_2O_3$  cathode, a metallic lithium anode, and a Celgard polypropylene separator. The electrolyte solution was 1M LiPF6 / DOL+DMC (1:1). The electrode was prepared by mixing 80 wt.% the hollow sphere  $c-In_2O_3$  sample with 10 wt.% acetylene black and 10 wt.% polypropylene fluoride (PVDF) binder in N-methylpyrrolidone (NMP) to form a slurry. The slurry was stirred 4h, then coated onto a copper foil, and dried at 120°C for 12h in a vacuum drier to prepare the cathode film. The cathode film was punched into discs (diameter=14mm). The cell was

fabricated in an Ar-filled glove box. The assembled cell was charged and discharged between 0.2V and 1.4V (versus Li/Li<sup>+</sup>) on the a multi-channel battery test system instrument at room temperature.

#### **3. RESULTS AND DISCUSSION**

Fig.1 shows a full set of diffraction peaks of the  $In_2O_3$  sample. The sharp peaks in the pattern showed good crystallinity of the  $In_2O_3$  sample. It is seen that a few of strong peaks at the near 21.49°, 30.58°, 35.47°, 51.04°, 83.20° can be indexed to the cubic phase of c- $In_2O_3$  (JCPDS no.06-0416), which is invisible in the XRD patterns[11]. The lattice parameters refined within this space group for the new phase are a=1.012nm (a=1.0118nm from JCPDS no.06-0416), and unit cell volume V<sub>0</sub>=0.065 nm<sup>3</sup>.



Figure 1. XRD pattern of hollow sphere c-In<sub>2</sub>O<sub>3</sub>

In Fig.2, it vividly reveals that the  $In_2O_3$  sample consists of hollow sphere microspheres and sphere-section with a diameter of about 1-3µm. More interestingly, both the inner and outer surface of the hollow c-In<sub>2</sub>O<sub>3</sub> sample are rough (Fig.2 a.b). The thickness of the hollow ball is about 0.6µm. It worth noting that some of the hollow ball is incomplete. Sphere wall was structure by nanocrystals (Fig.2 c). It is well-known that rate capability plays an important role in lithium ion battery. The electrochemical performance of hollow sphere c-In<sub>2</sub>O<sub>3</sub> were evaluated by galvanostatic measurements. Fig.3 shows that the first discharge curve of the coin cell between 0.2V and 1.4V, at a constant current density of 30 mA  $\cdot$ g<sup>-1</sup>.



**Figure 2.** (a) (b) SEM image of hollow sphere c-In<sub>2</sub>O<sub>3</sub>;(c) magnified SEM image of hollow sphere c-In<sub>2</sub>O<sub>3</sub> nanoparticles of the position marked in (b).

In the initial discharging process, there is a long smooth voltage plateaus around 0.9V, and The first discharge capacity is 1162 which is higher than that report by Zhou et al[2]. However, with the increase of the cycling times, the discharge capacity of the coin cell presents a downward trend. Especially, the second discharge capacity downs quickly from 1162 mAh·g<sup>-1</sup> to 201 mAh·g<sup>-1</sup>. After the five cycle, the discharge capacity curve of the coin cell keeps a stable condition.



Figure 3. The first discharge curve of the Li/In<sub>2</sub>O<sub>3</sub> coin cell



**Figure 4.** Electrochemical properties of hollow sphere c-In<sub>2</sub>O<sub>3</sub>: (a) specific capacity of Li/In<sub>2</sub>O<sub>3</sub> coin cell; (b) coulombic efficiency of Li/In<sub>2</sub>O<sub>3</sub> coin cell; (c) rate performance of Li/In<sub>2</sub>O<sub>3</sub> coin cell

It suggests that hollow sphere c-In<sub>2</sub>O<sub>3</sub> has large irreversible capacity. In the 40th cycle, the discharge capacity of the coin cell only remains about 100 mAh·g<sup>-1</sup> (Fig. 4a ). The massive capacity reduction may be attributed to lithiation reaction initially and delithiation of the electrode in the lithium ion battery [5]. Coulombic efficiency is a key index for evaluating the stability of the coin cell. It can be seen that the coulombic efficiency in the first cycle is low and unstable, and becames gradually increasing from 70% to 94% (Fig. 4b), which may be due to the diffusion rate of Li<sup>+</sup> ions in hollow sphere c-In<sub>2</sub>O<sub>3</sub> becoming stable. As shown in Fig. 3c, it can be seen that the discharge capacity of the coin cell decreased quickly at the current density of 30 mA·g<sup>-1</sup> during the first five cycles. When the current density gradually increased from 60 mA·g<sup>-1</sup> to 240 mA·g<sup>-1</sup>, the In<sub>2</sub>O<sub>3</sub> electrode delivered a capacity of 90 mAh·g<sup>-1</sup>, 82 mAh·g<sup>-1</sup>, 74 mAh·g<sup>-1</sup>, respectively. While the current density back to 30 mA·g<sup>-1</sup> after being continuously cycled for 10 times, a capacity of 101 mAh·g<sup>-1</sup> can be obtained, indicating good robustness and stability of hollow sphere c-In<sub>2</sub>O<sub>3</sub> material.

#### 4. CONCLUSION

In conclusion, the hollow sphere  $c-In_2O_3$  sample was prepared by hydrothermal method. According to the XRD and SEM, sample was cubic bixbyite-type structure (Ia-3)  $In_2O_3$ , good crystallinity and hollow sphere. The sample presented the large discharge capacity of 1162 mAh·g<sup>-1</sup> at the current density of 30 mA·g<sup>-1</sup> between 0.2V and 1.4V. The discharge capacity faded quikly in the charge-discharge process. After 40 cycles, the reversible capacity was around 100 mAh·g<sup>-1</sup>, and coulombic efficiency rises to 94%.

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## References

- 1. K.R. Prasad, K. Koga and N. Miura, Chem. Mater, 16 (2014) 1845.
- Y.N. Zhou, H. Zhang, M.Z. Xue, C.L. Wu, X.J. Wu and Z.W. Fu, J. Power Sources, 162 (2006) 1373.
- 3. D. Liu, W.W. Lei, S. Qin and Y. Chen, *Electrochim Acta*, 135 (2014) 128.
- 4. D. Liu, W.W. Lei, S. Qin, L.T. Hou, Z.W. Liu, Q.L. Cui and Y. Chen, *J Mater Chem A*, 1 (2013) 5274.
- 5. L. Zhao, W.B. Yue and Y. Ren, *Electrochim Acta*, 116 (2014) 31.
- 6. S. Qin, W.W. Lei, D. Liu, P. Lamb and Y. Chen, *Mater Lett*, 91 (2013) 5.
- 7. H. Li, X.J. Huang and L.Q. Chen, Solid State Ionics, 123 (1999) 189.
- 8. X.H. Huang, J.P. Tu, B. Zhang, C.Q. Zhang, Y. Li, Y.F. Yuan, et al. *J. Power Sources*, 161 (2006) 541.
- 9. W.H. Zhang and W.D. Zhang, J Solid State Chem, 186 (2012) 29.
- 10. B.X. Li, Y. Xie, M. Jing, G.X. Rong, Y.C. Tang and G.Z. Zhang, Langmuir, 22 (2006) 9380.
- 11. H.X. Yang, L. Liu, H. Liang, J.J. Wei and Y.Z. Yang, CrystEngComm, 13 (2011) 5011.
- 12. X. He, H.R. Liu, H.L. Dong, J. Liang, H. Zhang and B.S. Xu, J Inorg Mater, 29 (2014) 264.
- 13. X. Yang, J. Xu, T.L. Wong, Q.D. Yang and C.S. Lee, Phys Chem Chem Phys, 15 (2013) 12688.
- 14. M.M. Titirici, M. Antonietti and A. Thomas, Chem Mater, 18 (2006) 3808.
- 15. J. Yang, C.K. Lin, Z.L. Wang and J. Lin, Inorg Chemi, 45 (2006) 8973.
- 16. M.A. Mahdi, J.J. Hassan, S.J. Kasim, S.S. Ng and Z. Hassan, Mat Sci Semicon Proc, 26 (2014) 87.
- 17. Q.Y.Wang, K. Yu, F.Xu, J.Wu, Y.Xu and Z.Q. Zhu, Mater Lett, 62 (2008) 2710.
- 18. T.G. Kim, H.B. Oh, H. Ryu and W.J. Lee, J Alloy Compd, 612 (2014) 74.
- 19. R.Yang, J.Zheng, J.Huang, X.Z.Zhang, J.L.Qu and X.G.Li, Electrochem Commun, 12 (2010) 784.
- 20. W.H. Ho, C.F. Li, H.C. Liu and S.K. Yen, J. Power Sources, 175 (2008) 897.
- 21. Y.W. Hu, W. Zhang and W. Pan, Mater Res Bull, 48 (2013) 668.
- 22. Z.M. Li, P.Y. Zhang, T. Shao and X.Y. Li, Appl Catal B-Environ, 125 (2012) 350.
- 23. D. Liu, W. W. Lei, B. Zou, S. D. Yu, J. Hao, K. Wang, B. B. Liu, Q. L. Cui and G. T. Zou, *J. Appl. Phys.*, 104 (2008) 083506.

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