

Facile Synthesis of MoS₂ Modified TiO₂ Nanospheres with Enhanced Photoelectrocatalytic activity

Bin Dong^{1,2,*}, Yan-Ru Liu¹, Guan-Qun Han^{1,2}, Wen-Hui Hu¹, Yong-Ming Chai¹, Yun-Qi Liu^{1,*},
Chen-Guang Liu^{1,*}

¹ State Key Laboratory of Heavy Oil Processing, China University of Petroleum (East China), Qingdao 266580, PR China

² College of Science, China University of Petroleum (East China), Qingdao 266580, PR China

*E-mail: dongbin@upc.edu.cn, liuyq@upc.edu.cn, cgliu@upc.edu.cn

Received: 3 September 2015 / Accepted: 3 February 2016 / Published: 1 March 2016

MoS₂/TiO₂ nanocomposites composed of MoS₂ nanosheets and TiO₂ nanospheres have been successfully prepared by a facile hydrothermal process. The as-prepared MoS₂/TiO₂ samples with different MoS₂ content have been characterized by scanning electron microscopy (SEM), X-ray diffraction (XRD) and transmission electron microscopy (TEM). The results show TiO₂ nanospheres with uniform size can improve the dispersion and decrease the aggregation of MoS₂ nanosheets. The best morphology and size of MoS₂/TiO₂ nanocomposites can be obtained when the content of MoS₂ is 70 wt% (M-7). UV-vis data show that MoS₂/TiO₂ samples have better absorption in visible light region compared to pure MoS₂ and TiO₂. The photoelectrocatalytic activity of MoS₂/TiO₂ samples has been evaluated by the photocurrent measurement. The results show that MoS₂/TiO₂ nanocomposites with MoS₂ content of 70 wt% (M-7) have the highest photocurrent which implies best photoelectrocatalytic activity of M-7. The reason may be that the suitable content of MoS₂ and the tight junction between MoS₂ and TiO₂ nanospheres is helpful for preventing the recombination of photogenerated electrons and holes.

Keywords: TiO₂ nanospheres; MoS₂; photoelectrocatalytic activity; nanocomposites

1. INTRODUCTION

Due to its earth-abundant reserves and low cost, MoS₂ with excellent electronic and optical properties has attracted much interests on hydrotreating catalysis [1], lithium ion battery [2], electrocatalytic hydrogen evolution reaction [3], photocatalytic degradation of organic pollutants [4-7] and optoelectronic devices [8-9]. Some research has found that with the stacking layers of MoS₂ decreasing to monolayer, an increasing could be observed from indirect band gap (1.29 eV) to direct

band gap (1.90 eV), which implies the better absorption in the visible light range [10-11]. Therefore, more and more researchers have focused their effort on the photocatalytic and photoelectrochemical property of MoS₂ as the alternatives of expensive Pt-based electrocatalysts [12-14].

Recently, the theoretical and experimental research has proved that active sites of layered MoS₂ are on the Mo-edges sites and the unsaturated sulfur atoms [15-16]. However, the 2D layered structure makes MoS₂ facile to aggregate and inefficiently adsorb reactant molecular because of its low specific surface area. Furthermore, the more stacking layers of MoS₂ on 2D nanosheets, the less active sites. As a result, the catalytic activities dramatically decrease. Therefore, exposing more edges and rims of MoS₂ nanosheets can improve the catalytic activities. Many researches have been done to prepare novel nanostructure of MoS₂ [17-18] or hybrid nanomaterials [19-22] to increase the catalytic activity sites of MoS₂. Other than the number of active sites, the band gap of MoS₂ is also a key factor for photoelectrochemical HER. Owing to monolayer of MoS₂ has best absorption in visible light range, controlling the stacking of MoS₂ nanosheets is our goal.

In our work, MoS₂/TiO₂ nanocomposites have been synthesized via a simple hydrothermal reaction. TiO₂ nanospheres with uniform size and high area surface have been used as support to improve the dispersion and restrict the stacking numbers of MoS₂ nanosheets. The UV-vis and photocurrent measurements of the samples have been investigated in detail. The as-prepared MoS₂/TiO₂ nanocomposites exhibit higher photoelectrocatalytic activity for hydrogen evolution reaction.

2. EXPERIMENTAL SECTION

2.1 Preparation of TiO₂ nanospheres

The preparation of TiO₂ nanospheres references to a facile process reported by the literature [23]. The detailed method is as following: firstly, 5 ml tetrabutoxytitanium was added into 50 ml ethylene glycol ((CH₂OH)₂), and the mixture solution was stirred for 8 h. Then 200 ml acetone was poured into the former mixture solution under stirring for 1h. The obtained white precipitates were washed and dried at 60 °C overnight. Then titanium glycolate precursor was obtained. Next, 0.300 g titanium glycolate precursor was added into 60 ml deionized water, and refluxed for 1h at 80 °C. The white precipitates were washed and dried at 60 °C overnight. Finally the white TiO₂ nanospheres were obtained.

2.2 Preparation of MoS₂/TiO₂ nanocomposites

MoS₂/TiO₂ nanocomposites (70 wt% of MoS₂, denoted as M-7) were prepared via a hydrothermal method. Firstly, 0.06 g sodium molybdate (NaMoO₄·2H₂O) and 0.12 g thiourea (C₂H₅NS) were dispersed in 70 ml deionized water. Then 0.013 g TiO₂ nanospheres were added into the above solution and stirred for 0.5 h. The obtained suspension was transferred into a Teflon-lined stainless steel autoclave and then heated in an electric oven at 220 °C for 24 h. The black precipitates

were washed and dried at 50 °C overnight. Following the above mentioned procedure, MoS₂/TiO₂ nanocomposites with different content (50 wt%, 90% of MoS₂, denoted as M-5 and M-9) were also prepared. For comparison, the pure MoS₂ was prepared under the identical conditions.

2.3 Photoelectrochemical tests

The photoelectrochemical characterization of the different samples have been measured by electrochemical workstation (CHI 604E, Chenhua, China) with a three-electrode composed of FTO electrode decorated by different MoS₂/TiO₂ nanocomposites as working electrode, platinum foil as the auxiliary electrode, and a silver chloride electrode (Ag/AgCl) as a reference electrode. The working electrodes were prepared as follows: 20 mg of the samples were dispersed in 2 ml absolute ethanol of Nafion to obtain samples slurry. Then, the slurry was dropped onto a 1.5 cm×1.0 cm F-doped SnO₂-coated glass (FTO glass) electrode. Next, these electrodes were dried in an oven at 60 °C for 3 h. All the measurements were carried out in the aqueous solution of 0.35 M/0.25 M Na₂S-Na₂SO₃ without bias potential at room temperature. A 100 W xenon lamp with visible light wavelength range has been used as light source with 50 mW/cm² incident light intensity to measure the photoelectrochemical properties.

2.4 Characterization

Crystallographic structure of all as-prepared samples was investigated with X-ray powder diffraction (XRD, X'Pert PRO MPD, Cu KR) at a scanning rate of 1 °C min⁻¹. XRD data were collected in the 2θ ranges from 5 to 76°. The morphology of the samples was examined with field-emission scanning electron microscopy (SEM, Hitachi, S-4800). Transmission electron microscopy (TEM) images were collected on HRTEM, JEM-2100UHR with an accelerating voltage of 200 kV. The samples were prepared by dropping the ethanol solution of samples on the Cu grids. UV-vis spectra were recorded on a UV-vis spectrometer (UV-2600, Shimadzu, Japan) over a spectral range of 200-1000 nm.

3. RESULTS AND DISCUSSION

X-ray diffraction (XRD) patterns of pure MoS₂, M-7 and TiO₂ nanospheres are shown in Figure 1. As for pure MoS₂ sample (curve a in Figure 1), the (002), (100), (103), (110) planes corresponding peaks can be assigned to 14.1°, 32.9°, 39.5° and 58.8°(JCPDS 01-075-1539), respectively.

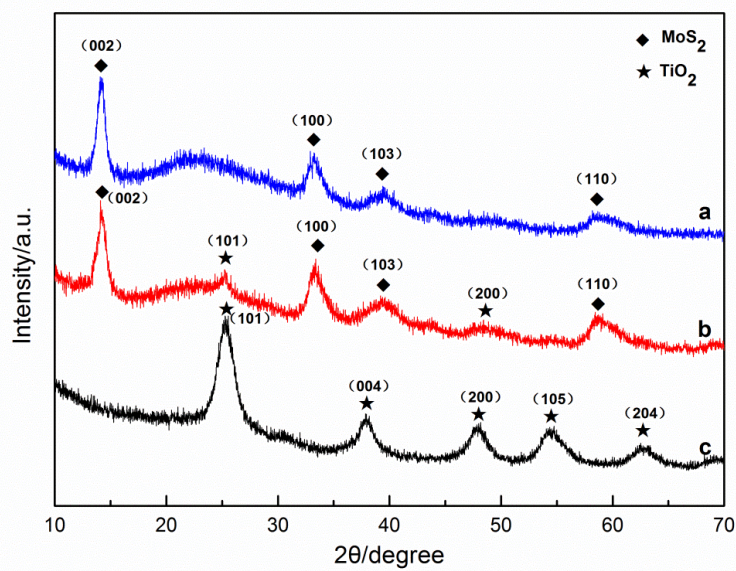
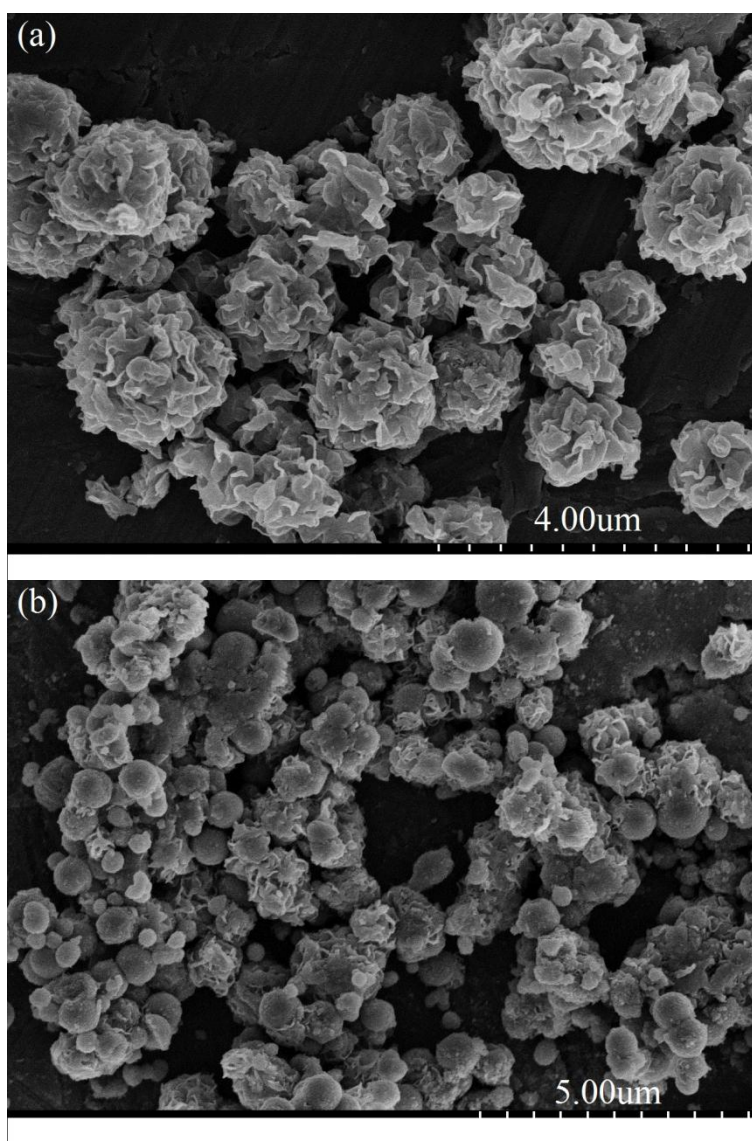


Figure 1. XRD data of the different samples: (a) pure MoS₂; (b) MoS₂/TiO₂ (M-7) and (c) pure TiO₂.



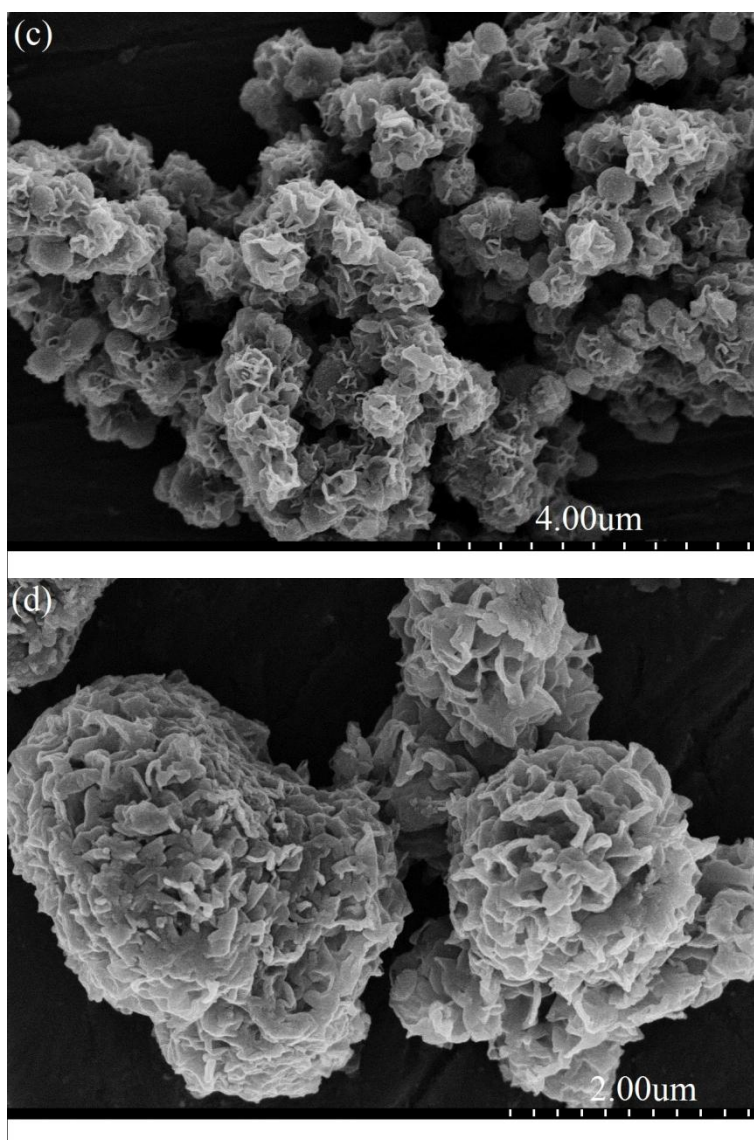
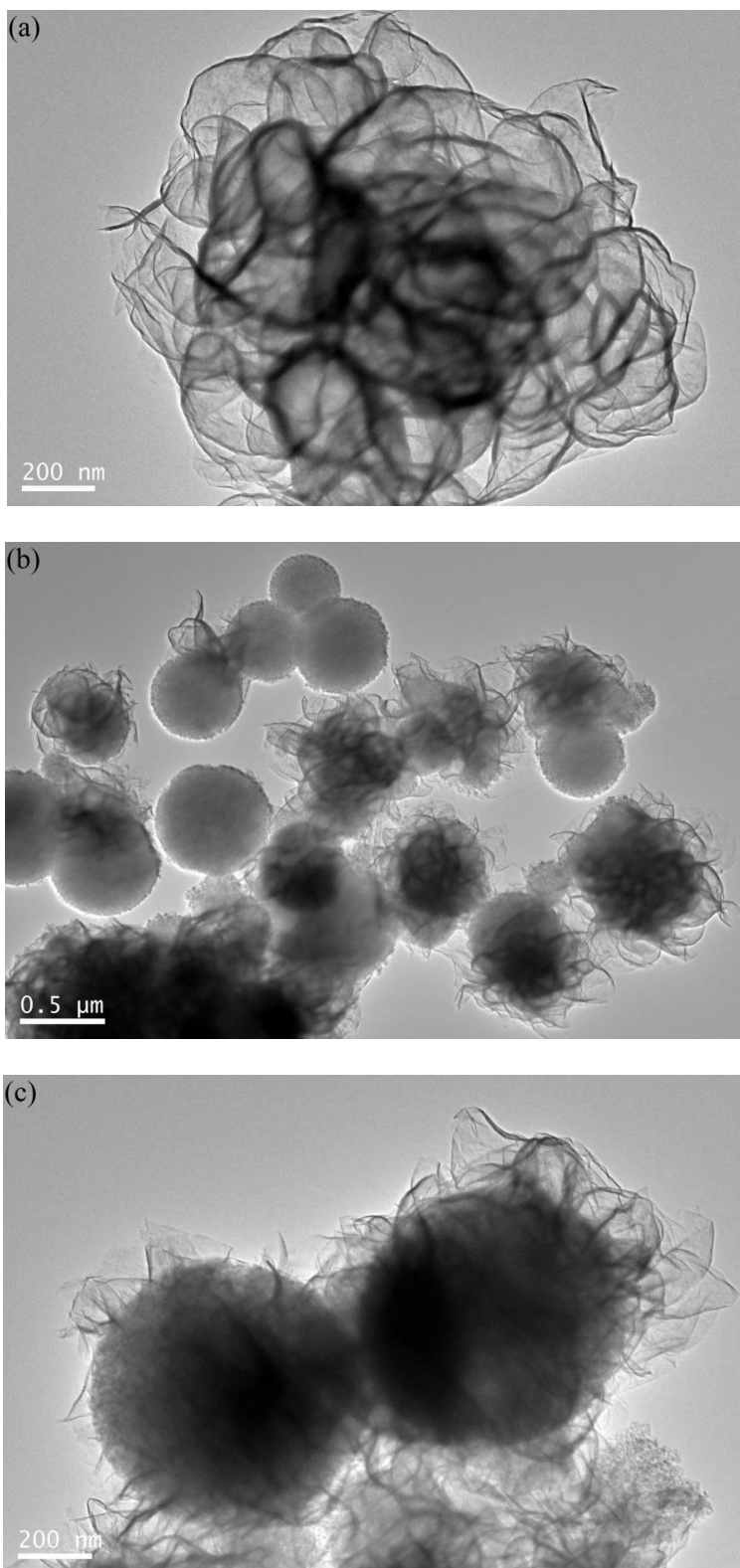


Figure 2. SEM images of the different samples: (a) pure MoS₂; (b) MoS₂/TiO₂(M-5); (c) MoS₂/TiO₂ (M-7); (d) MoS₂/TiO₂ (M-9).

It can be seen that the peak at 14.1° become weaker in M-7 (curve b in Figure 1) than that in pure MoS₂, indicating that TiO₂ nanospheres restrict the growth along (002) plane of MoS₂. The stacking layers of MoS₂ will decrease with the weakness of (002), which implies that the MoS₂ in M-7 can expose more edges and rims as catalytic sites. The diffraction peaks of the pure TiO₂ nanospheres (curve c in Figure 1) are in keeping with the peaks of the standard pure anatase TiO₂ (JCPDS: 01-071-1167). The broader peaks of (101), (004), (200), (204) indicate that TiO₂ nanospheres have the amorphous state and low crystalline.

Figure 2a displays the corresponding SEM image of pure MoS₂. The pure MoS₂ obtained by the same hydrothermal method have the nanoflower morphology of with the severe stacking and large size of the diameter of 2 μm. SEM image of M-5 (with 50 wt% of MoS₂) is shown in Figure 2b. Absolutely, only few TiO₂ nanospheres can be decorated by MoS₂. The MoS₂/TiO₂ nanocomposites

can't be successfully synthesized when adding 50 wt% of MoS_2 . Compared to M-5, TiO_2 nanospheres in M-7 with the diameter of 0.5 μm were mostly packaged by MoS_2 , and the nanocomposites of M-7 exhibit the spherical shape with good distribution and similar size (in Figure 2c).



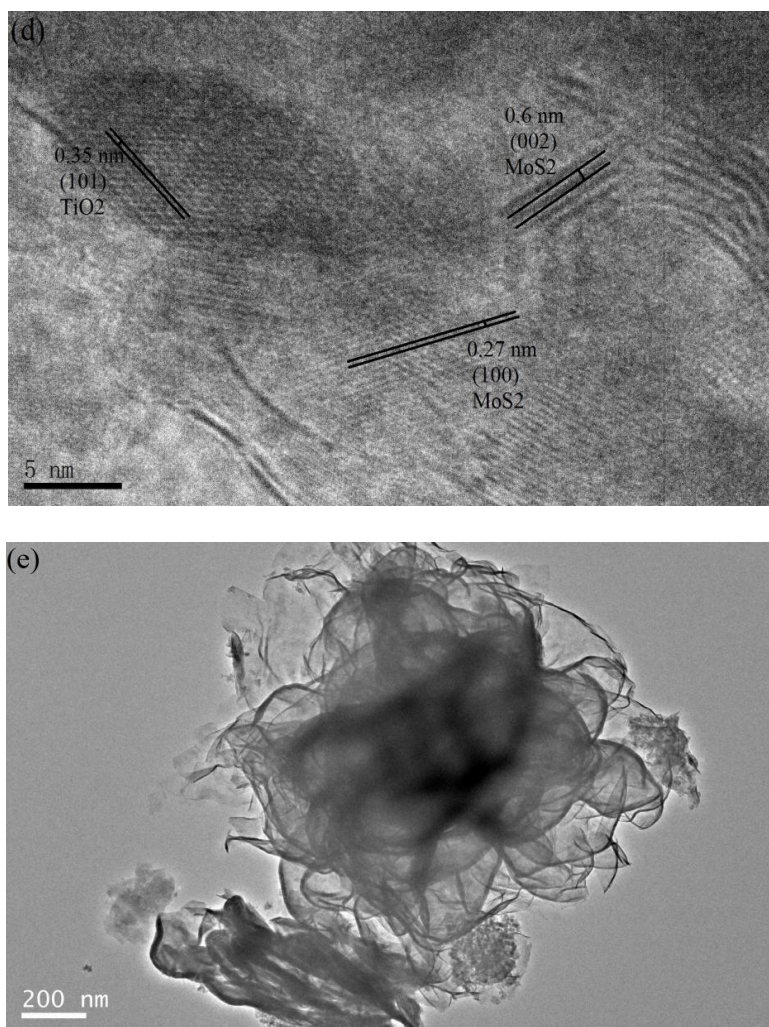


Figure 3. TEM images of the different samples: (a) pure MoS₂; (b) MoS₂/TiO₂(M-5); (c) MoS₂/TiO₂ (M-7); (d) HRTEM of M-7; (e) MoS₂/TiO₂ (M-9).

It can be seen that MoS₂ homogenously disperse on the surface of TiO₂ nanospheres which means that the MoS₂/TiO₂ composites are prepared successfully. When increasing the content of MoS₂ to 90% in M-9, the obvious stacking of MoS₂ sheets can be observed in Figure 2d. The diameter of M-9 also increases to about 2 μm , which may be caused by the excess of MoS₂. Therefore, M-7 have the better morphology and junction between MoS₂ and TiO₂.

Transmission electron Microscopy (TEM) images of the different samples are displayed in Figure 3. Figure 3a shows the typical stacking morphology of pure MoS₂ nanosheets. Obviously, pure MoS₂ grow rapidly and have severe aggregation in the absence of TiO₂ nanospheres. When adding TiO₂ nanospheres to synthesize M-5, only few MoS₂/TiO₂ nanocomposites can be seen in Figure 3b, which is identified by SEM image in Figure 2b. Figure 3c shows the TEM image of M-7, indicating that the few layers of MoS₂ nanosheets can homogenously grow on the surface of TiO₂ nanospheres with the small diameter of 0.5 μm . A high-resolution TEM (HRTEM) image (Figure 3d) also revealed the tight recombination between MoS₂ and TiO₂. Under higher resolution, the layered structure of MoS₂ nanosheets with an average spacing of 0.6 nm can be seen, which belongs to the (002) facet of MoS₂. The crystallographic spacing of 0.27 nm corresponds to the lattice parameter in the (100) plane

of MoS_2 . Furthermore, the lattice spacing of 0.35 nm is equal to the (101) plane of TiO_2 . These results proved that the good junction of MoS_2 and TiO_2 can be obtained when the content of MoS_2 is 70 wt%. When the content of MoS_2 is up to 90%, the severe stacking of MoS_2 nanosheets and the larger size can be observed in Figure 3e.

The structure and properties of photoelectrocatalysts related to bandgap, size, and/or band position has important effect on the absorption properties. UV-vis absorption measurement has been used to evaluate bandgaps of the different samples.

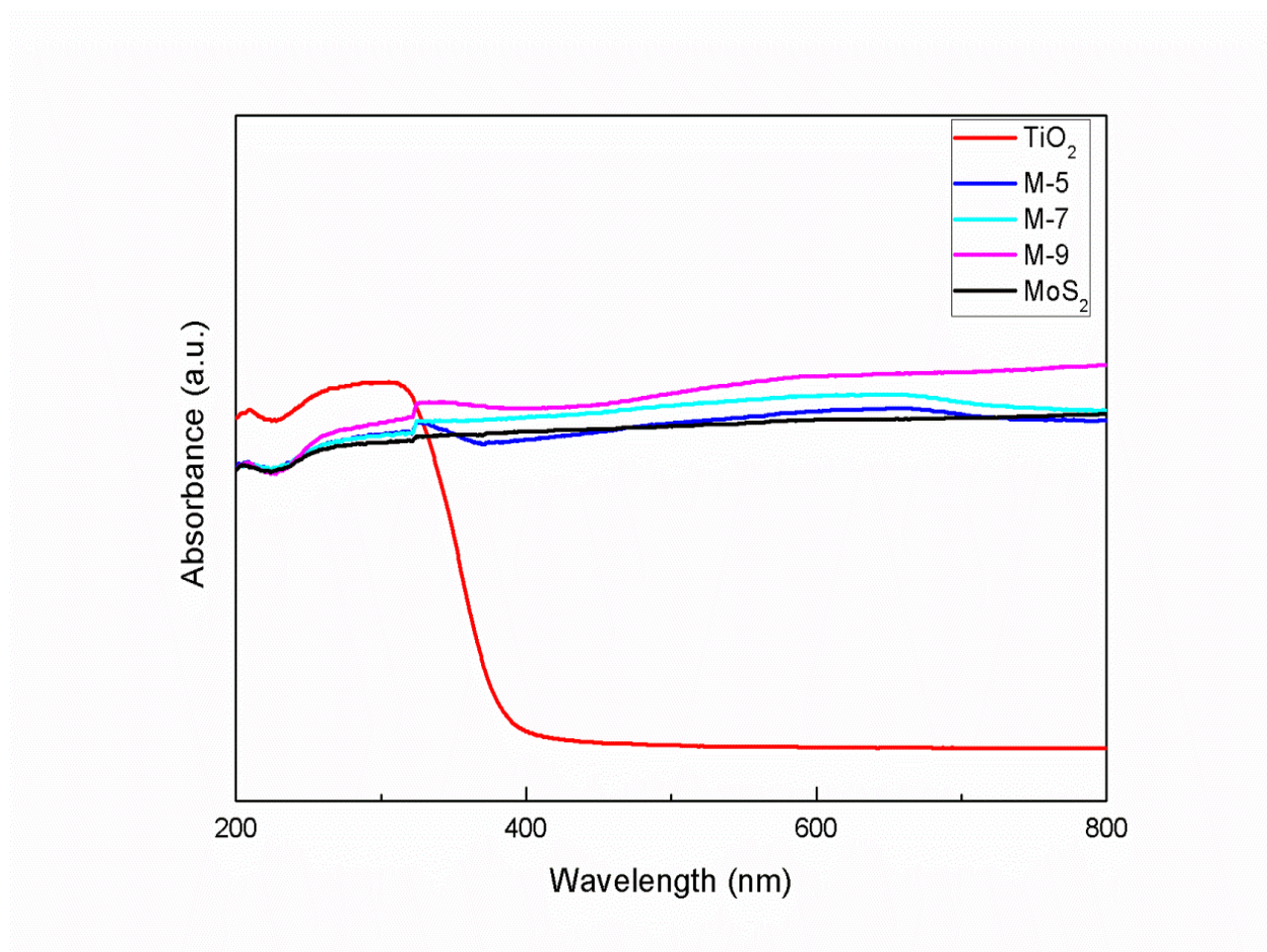


Figure 4. UV-vis absorption spectra of TiO_2 nanospheres; $\text{MoS}_2/\text{TiO}_2$ (M-5); $\text{MoS}_2/\text{TiO}_2$ (M-7); $\text{MoS}_2/\text{TiO}_2$ (M-9) and pure MoS_2 .

As shown in Figure 4, the pure TiO_2 nanospheres only have the basic absorption band in UV light range. Either pure MoS_2 or $\text{MoS}_2/\text{TiO}_2$ nanocomposites show enhanced absorption in visible light region. Compared with the TiO_2 nanospheres' calculated bandgap energy (3.20 eV), the bandgap energy of $\text{MoS}_2/\text{TiO}_2$ nanocomposites is close to 2.80 eV. This proves the as-prepared $\text{MoS}_2/\text{TiO}_2$ nanocomposites is very suitable for photoelectrocatalytic HER [24].

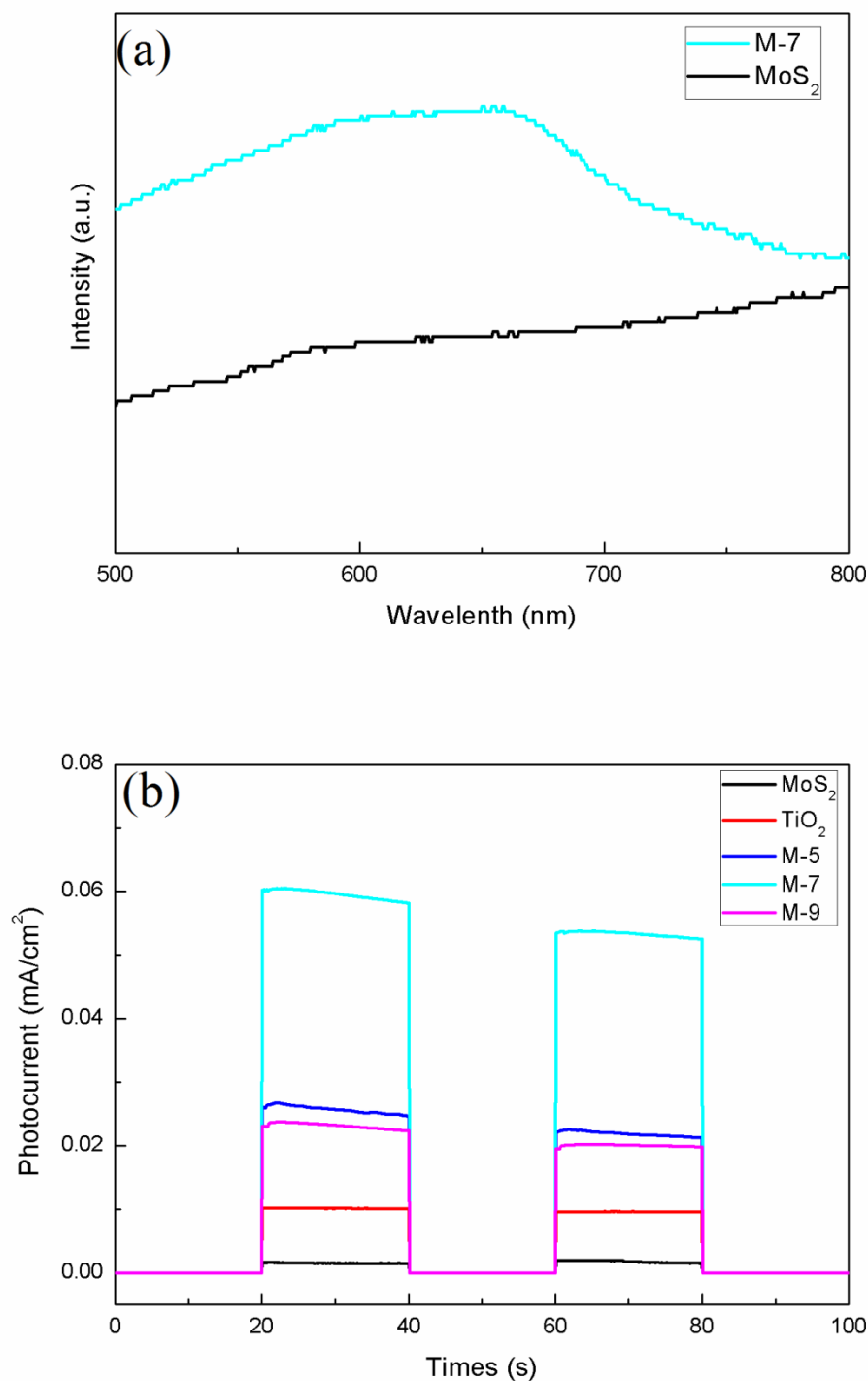


Figure 5. (a) Photoluminescence spectra of pure MoS_2 and $\text{MoS}_2/\text{TiO}_2$ (M-7); (b) The photocurrent measurements of TiO_2 nanospheres; $\text{MoS}_2/\text{TiO}_2$ (M-5); $\text{MoS}_2/\text{TiO}_2$ (M-7); $\text{MoS}_2/\text{TiO}_2$ (M-9) and pure MoS_2 .

It has been proved that MoS_2 monolayer has the strongest photoluminescence (PL) absorption compared to the bulk MoS_2 [25]. A broad peak centered at 650 nm (1.92 eV) can be observed on $\text{MoS}_2/\text{TiO}_2$ nanocomposites in Figure 5a. The band gap energies of about 1.9 eV proved the existence

of MoS₂ monolayer structure [26]. In addition, the absorption intensity of MoS₂/TiO₂ nanocomposites is higher than that of pure MoS₂, indicating well dispersion of MoS₂ in the nanocomposites than in the pure MoS₂.

In order to characterize the different samples' ability of separating of photogenerated electrons and producing charge carriers, the photocurrent measurements are used. As shown in Figure 5b, M-7 shows the strongest photocurrent intensity than pure MoS₂, TiO₂ and M-5, M-9. Therefore, the results imply that M-7 has better structure and tends to easier produce charge carriers and separated electrons. Because the presence of photogenerated electrons is the key factor in actual photocatalytic HER performance, MoS₂ in M-7 can transfer more photogenerated electrons to protons for improving hydrogen evolution reaction. So M-7 is expected to exhibit enhanced photocatalytic activity for HER.

4. CONCLUSIONS

Novel MoS₂/TiO₂ nanocomposites with well dispersed MoS₂ nanosheets on the surface of TiO₂ nanospheres have been successfully synthesized via a facile hydrothermal reaction. The morphology and structure of MoS₂/TiO₂ nanocomposites have been characterized. The loading of MoS₂ can impact the dispersion and junction of the nanocomposites. The M-7 with the 70 wt% content of MoS₂ has the best dispersion and heterostructure, which is helpful to improve the photocatalytic H₂ production activity. UV-vis and the photocurrent tests also proved the best photoelectrochemical activity of M-7. The unique heterostructure composed of MoS₂ and TiO₂ nanospheres break new grand in designing newfangled hierarchical MoS₂-based photoelectrocatalysts for HER.

ACKNOWLEDGEMENTS

This work is financially supported by the National Natural Science Foundation of China (U1162203 and 21106185) and the Fundamental Research Funds for the Central Universities (15CX05031A).

References

1. R.R. Chianelli, G. Berhault and B. Torres, *Catal. Today*, 147 (2009) 275.
2. J. Cho, H. Hwang and H. Kim, *Nano Lett.*, 11 (2011) 4826.
3. C.G. Morales-Guio, L. A. Stern and X.L. Hu, *Chem. Soc. Rev.*, 43 (2014) 6555.
4. Q. Xiang, J. Yu and M. Jaroniec, *J. Am. Chem. Soc.*, 134(15) (2012) 6575.
5. F.A. Frame and F. E. Osterloh, *J. Phys. Chem. C*, 114(23) (2010) 10628.
6. W.J. Zhou, Z.Y. Yin, Y.P. Du, X. Huang, Z.Y. Zeng, Z.X. Fan, H. Liu, J.Y. Wang and H. Zhang, *Small*, 9 (2013) 140.
7. H. Zhao, Y. Dong, P. Jiang, H. Miao, G. Wang and J. Zhang, *J. Mater. Chem. A*, 3(14) (2015) 7375.
8. H. S. Lee, S. W. Min, Y. G. Chang, M. K. Park, T. Nam, H. Kim, J. H. Kim, S. Ryu and S. Im, *Nano Lett.*, 12 (2012) 3695.
9. O. Lopez-Sanchez, D. Lembkel, M. Kayci, A. Radenovic and A. Kis, *Nat. Nanotechnol.*, 8 (2013) 497.
10. T.S. Li and G.L. Galli, *J. Phys. Chem. C*, 111 (2007) 16192.
11. C. Lee, H. Yan, L.E. Brus, T. F. Heinz, J. Hone and S. Ryu, *ACS Nano*, 4 (2010) 2695.

12. G. J. Zhou, Y. M. Xu, Y. Z. Fu, Y. Yang and Y. C. Zhang, *Int. J. Electrochem. Sci.*, 9 (2014) 3990.
13. Y. Z. Song, W. H. Song, L. Qian, J. H. Wang, X. M. Zhang, X. M. Lv and J. M. Xie, *Int. J. Electrochem. Sci.*, 9 (2014) 6843.
14. Y. G. Li, H. L. Wang, L. M. Xie, Y. Y. Liang, G. S. Hong and H. J. Dai, *J. Am. Chem. Soc.*, 133 (2011) 7296.
15. B. Hinnemann, P. G. Moses, J. Bonde, K. P. Jørgensen, J. H. Nielsen, S. Hørch, I. Chorkendorff and J. K. Nørskov, *J. Am. Chem. Soc.*, 127 (2005) 5308.
16. T. F. Jaramillo, K. P. Jørgensen, J. Bonde, J. H. Nielsen, S. Hørch and I. Chorkendorff, *Science*, 317 (2007) 100.
17. M. Remskar, A. Mrzel, Z. Skraba, A. Jesih, M. Ceh, J. Demsar, P. Stadelmann, F. Levy and D. Mihailovic, *Science*, 292 (2001) 479.
18. G. Nagaraju, C. Tharamani, G. Chandrappa and J. Livage, *Nanoscale Res. Lett.*, 2 (2007) 461.
19. Q. Xiang, J. Yu and M. Jaroniec, *J. Am. Chem. Soc.*, 134 (2012) 6575.
20. S. Min and G. Lu, *J. Phys. Chem. C*, 116 (2012) 25415.
21. C. Liu, L. Wang, Y. Tang, S. Luo, Y. Liu, S. Zhang, Y. Zeng and Y. Xu, *Appl. Catal. B*, 164 (2015) 1.
22. Meng Shen, Zhiping Yan, Lei Yang, Pingwu Du, Jingyu Zhang and Bin Xiang, *Chem. Commun.*, 50 (2014) 15447.
23. J. Brivio, D.T. Alexander and A. Kis, *Nano Lett.*, 11 (2011) 5148.
24. M. Weber and M. Dignam, *Int. J. Hydrogen Energy*, 11 (1986) 225.
25. J.Z. Ou, A.F. Chrimes, Y. Wang, S.Y. Tang, M.S. Strano and K. Kalantar-Zadeh, *Nano Lett.*, 14 (2014) 857.
26. E.S. Kadantsev and P. Hawrylak, *Solid State Commun.*, 152 (2012) 909.