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Improving Power Conversion Efficiency of P3HT/PCBM based Organic Solar Cells by Optimizing Graphene Doping Concentration and Annealing Temperature

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Graphene (GN) doping concentration and annealing temperature in blended poly (3-hexylthiophene) and [6,6]-phenyl C61 butyric acid methyl ester (P3HT/PCBM) with GN (P3HT/PCBM/GN) based organic solar cells (OSC) were investigated. Results indicate the power conversion efficiency (PCE) of P3HT/PCBM /GN based OSC can be efficiently increased by 3 wt% GN doping and 120°C annealing. The OSC doped with 3 wt% GN exhibited the highest PCE among OSC doped with 0~10 wt% GN under the same annealed condition: unannlealed status and 120°C annealing. The PCE of 120°C annealed OSC doped with 3 wt% GN is 2.81, which is almost two times higher than that doped with 10 wt% GN. Thermal desorption spectra of unannealed P3HT/PCBM /3 wt% GN show two desorption peaks at around 130 and 182°C resulting from water vapor. The two temperatures are close to the glass transition temperature of PCBM and melting temperature of P3HT, respectively. The 120°C annealing on P3HT/PCBM/3 wt% GN can make both P3HT and PCBM have relatively good crystallization compared with unannealed status and 170°C annealing. Good crystallization makes charge carriers transport easily in P3HT/PCBM/3 wt %GN based OSC.

Keywords: power conversion efficiency, P3HT/PCBM, grapheme

1. INTRODUCTION

Organic solar cells (OSC) are economical and easily large-area prepared compared with compound-based and thin-film solar cells. Bulk heterojunction solar cells made by blending p-type conjugated polymers with n-type conjugated polymers, fullerenes, fullerene derivatives or nanoparticles are the most popular one in OSC. They are composed with a chain of nano-sized area between electron donors and acceptors in OSC [1-4]. Among numerous photoactive donor/acceptor composites, the p-type conjugated polymers with fullerene derivatives like poly (3-hexylthiophene)

(P3HT) and [6,6]-phenyl C61 butyric acid methyl ester (PCBM) blend have been intensively investigated[5,6]. The OSC composed of π -conjugated polymers and fullerene derivatives have attracted considerable interest due to their lightness, simple solution processing and bending capability.[7,8] The power conversion efficiency (PCE) of OSC can be improved by doping nanomaterials like carbon nanotubes [9,10], nanodiamonds [11], Fe3O4 magnetic nanoparticles [12], and ZnO nanorods [13]. Nanomaterials, which can be used to make super strong metals [14], have applied in solar cells due to their high surface to volume ratio, quantum confinement and quantum size effect [15]. Our group has doped Si nanoparticles, CdSe core-shell nanoparticles and nano-diamonds respectively into OSC, and the results indicate the PCE of OSC can be all improved with an upper limit of doping concentration.[16-18].

With superior electron transport properties, graphene (GN) has been well adjusted in OSC as both the electron acceptor material and the transparent electrode [19, 20]. However, their efficiency is extremely confined as compared with fullerene derivatives by reason of their conductor characteristics, which interrupt the effective hole mobility and aid the formation of recombination pathways [21].

Doping nanomaterials into photoactive layer can make positive effects such as increasing interfaces between donors and acceptors, optical absorption on photovoltaic performance. However, negative effects may also arise when nanomaterials are doped into OSC. Doping nanoparticles into P3HT/PCBM based OSC decreases the performance of OSC due to quenching of the excited state in the polymer. Segregation phenomena in OSC due to over doping have been reported [22]. Our experiences in doping nanomaterials into P3HT/PCBM reveal that the existence of the optimum doping concentration corresponds to PV characteristics of processed OSC[16-18].

Thermal annealing can provide benefits on photovoltaic process of P3HT/PCBM based OSC, such as improvement in surface morphology and crystallinity of photoactive materials, furthermore contact resistance between active materials and electrode to aid charge generation, transport and collection [23]. However, thermal annealing can also bring problems on photovoltaic process of P3HT/PCBM based OSC if thermal annealing is too much, like PCBM aggregation, phase separation between P3HT and PCBM to lower PCE [24]. Selecting annealing temperature is one of the key factors in improving short circuit current density (J_{sc}), open circuit voltage (V_{oc}) and PCE [25]. Selecting optimum annealing temperature of nanodiamonds doped P3HT/PCBM has been studied with thermal desorption spectroscopy in our group [18]. The results indicate 120°C is the optimum annealing temperature for nanodiamonds doped P3HT/PCBM based OSC.

This paper investigates influence of GN doping concentration and annealing temperature in processing P3HT/PCBM/GN based OSC on photovoltaic properties. Photovoltaic characteristics of the P3HT/PCBM based OSC with different GN doping concentration were measured first. The obtained OSC with the optimum GN concentration were annealed with different annealing temperature. The optimized GN doping concentration and annealing temperature to produce P3HT/PCBM/GN based OSC were obtained in this work.

2. EXPERIMENTS

The hybrid OSC were fabricated on indium tin oxide (ITO)-coated glass substrates. Designed ITO substrates were ultrasonically washed, and then oxygen plasma processed in a reactive ion etching

tool raise hydrophilic property of the ITO surface. А poly(3,4to layer of ethylenedioxythiophene)/poly(styrenesulfonate) (PEDOT/PSS, CLEVIOS™ P VP AI 4083, Heraeus Clevios GmbH, Leverkusen, Germany) about 40 nm thick was spin-coated on the ITO substrate and baked at 120 °C for 30 min. The photoactive layer composed P3HT (RMI-001E, Rieke Met. Inc., Lincoln, Nebraska, USA) and PCBM ([60]PCBM, Nano-C, Westwood, Maryland, USA) in 10:8 wt % ratio dissolved in 1,2-dichlorobenzene with 0, 1, 3, 5, 10 wt% GN. Some samples were annealed at 120°C according to our previous optimizing results on other PCBM/P3HT based OSC [16-18]. Lastly, a Ca/Al electrode about 120 nm thick was thermally evaporated on the active layer.

The device structure of the produced OSC, in which P3HT/PCBM/GN is the photoactive layer of the device, was shown in figure 1.



Figure 1. Device structure of P3HT/PCBM/Graphene based photovoltaic device

Raman and optical absorption spectra of processed samples were measured by Micro-Raman spectroscopy (Ulvac, Jobin-Yvon) and optical spectrometer (U-4100, Hitachi, Minato-ku, Tokyo Japan) respectively. The current–voltage measurements were carried by using a source meter (2410, Keithley, Cleveland, Ohio, USA), and a solar simulator (Sun 3000 class AAA, TELTEC, Hong Kong) with the AM 1.5 filter under an irradiation intensity of 100 mW cm⁻². The measured photovoltaic characteristics of processed OSC with varied GN concentrations indicate OSC with 3 wt% GN demonstrates best PV characteristics for both unannealed and 120°C annealed photoactive materials. The thermal desorption spectra (TDS) of unannealed P3HT/PCBM/3 wt% GN films were measured by a self-made measurement system, which is described in detail in reference 18. Only one apparent mass signal corresponding to water vapor was observed. Two desorption peaks in TDS were observed at about 130 and 182°C, similar to previous results on P3HT/PCBM/4 wt %ND films [18]. The annealing temperature of P3HT/PCBM/3 wt %GN films was selected at 120 and 170°C respectively (about 10°C lower than the corresponding temperatures of two desorption peaks). Optical absorption spectra and

photovoltaic properties of P3HT/PCBM/3 wt% GN annealed at 120 and 170°C based OSC were estimated.

3. RESULTS AND DISCUSSION

The measured Raman spectra of OSC with GN doping show an apparent peak corresponding to GN. Figure 2 shows two spectra corresponding to OSC undoped and doped with 3 wt% GN respectively. The peak at about 2700 cm⁻¹ corresponds to G' band of graphene resulted from OSC doped with 3 wt% GN in figure 2 [26]. The peak at about 1442 and 1380 cm⁻¹ can be attributed to C=C stretching vibrations of the thiophene ring and C-C skeletal stretching in P3HT respectively [27].



Figure 2. The measured Raman spectra corresponding to OSC undoped and doped with 3 wt% GN



A



Figure 3. Photovoltaic performance: a. open circuit voltage (Voc), b. short circuit current density (Jsc), c. fill factor (FF) and d. power conversion efficiency(PCE) of unannealed and 120°C annealed P3HT/PCBM/GN with different GN doping concentrations

Photovoltaic performance: open circuit voltage (Voc), short circuit current density (Jsc), fill factor (FF) and power conversion efficiency (PCE) of unannealed and 120°C annealed P3HT/PCBM/GN based OSC with different GN doping concentrations are shown in figure 3a, 3b, 3c

and 3d respectively. The OSC with 3 wt% GN demonstrates the highest value of all four parameters in figure 3 for both unannealed and 120°C annealed OSC. The PCE of 120°C annealed OSC rises from 2.24 to 2.81% with GN doping concentration till 3 wt%. The GN doping raises the donor/acceptor interface area for charge separation and reduces the electron transfer path in P3HT/PCBM. However, over 3 wt% GN doping provides problems like segregation phenomena in polymer [22]. The PCE of 120°C annealed OSC with 10 wt% GN doping is almost two times lower than that with 3 wt% GN doping observed from figure 3d. The Jsc, FF and PCE of 120°C annealed OSC is higher than those of unannealed OSC for the same GN doping concentration observed from figure 3b , 3c and 3d. Annealing can improve nanoscale morphology and crystallinity of photoactive layer. Photovoltaic performance is therefore improved by thermal annealing.



Figure 4. Thermal desorption spectroscopy of unannealed P3HT/PCBM/3 wt %GN films

The TDS of unannealed P3HT/PCBM/3 wt% GN films was displayed in figure 4. Only one desorption species clearly observed from TDS measurement system corresponds to ion mass to charge ratio: 18. The species attributes water vapor. This may relate with significantly hygroscopic PEDOT:PSS and PCBM contained in measured OSC [28-29]. Two desorption peaks at around 130 and 180°C in figure 4 similar to those we investigated on OSC whose photoactive materials are P3HT/PCBM/ doped with nano diamonds [18]. The two temperatures attribute to the reported glass transition temperature of PCBM (131.2°C) and melting temperature of P3HT (178°C) [30, 31]. These two desorption peaks in figure 4 correspond to two activated states connected with glass transition temperature of PCBM and melting temperature of P3HT/PCBM/3 wt % GN.

Optical absorption spectra of P3HT/PCBM/3 wt% GN annealed at different annealing temperatures (unannealed, 120, 170°C) were shown in figure 5. The peak located at 300~400 nm attributes to PCBM. Its intensity reduces with annealing temperature. The other peak located at 500~600 nm attributes to P3HT and its intensity grows with annealing temperature up to 120°C. This phenomenon may attribute to crystallization of photoactive materials and PCBM segregation.



Figure 5. Optical absorption spectra of P3HT/PCBM/3 wt% GN annealed at different annealing temperatures (unannealed, 120, 170°C).

The intensity of optical absorption in figure 5 relates with crystallization of photoactive materials. PCBM crystallizes faster than P3HT, and can homogeneously grow crystals at room temperature [32]. Annealing enhances the crystallization of P3HT and disconnection of PCBM in P3HT/PCBM. PCBM disconnects and constructs isolated clusters as its crystals start to break-up during annealing process [33].

Table 1. Photovoltaic characteristics	of the P3HT/PCBM/3 wt	t% GN based OSC anne	aled at different
temperatures (unannealed, 120	, 170°C).		

Photovoltaic Characteristics Annealing status	Voc (V)	Jsc (mA/cm ⁻²)	FF (%)	PCE (%)
Unannealed	0.619	6.84	55.83	2.36
120°C annealing	0.619	7.05	64.30	2.81
170°C annealing	0.599	4.63	68.16	1.89

Photovoltaic characteristics and current density-voltage (J-V) characteristics of the P3HT/PCBM/3 wt% GN based OSC annealed at different temperatures were shown in table 1 and figure 6 respectively.



Figure 6. Current density-voltage (*J*–*V*) characteristics of the P3HT/PCBM/3 wt% GN based OSC annealed at different temperatures (unannealed, 120, 170°C).

The magnitude sequence of PCE for OSC in table 1 is: 120° C > unannealed > 170° C. The first desorption peak at 130° C in figure 3 corresponds to a turning point in P3HT/PCBM based OSC. The 130° C peak corresponds to glass transition temperature of PCBM. Crystallization of P3HT and PCBM is critical for photovoltaic characteristics of the P3HT/PCBM based OSC. Well crystallized P3HT/PCBM enhances charge carriers move easily in photoactive layer during photovoltaic process. Annealing at 120° C is close but lower than the turning point : 130° C can cause P3HT as well as PCBM in P3HT/PCBM/3 wt %GN have comparatively good crystallization seen from the trend of absorption intensity in figure 5.

4. CONCLUSIONS

An obvious peak at about 2700 cm⁻¹ in Raman spectra corresponds to G' band of graphene resulted from OSC with GN doping. The PCE of P3HT/PCBM/GN based OSC can be improved by adjusting GN doping concentration and annealing temperature on photoactive layer. The P3HT/PCBM/3 wt% GN based OSC GN demonstrates the highest value of Voc, Jsc, FF and PCE compared to OSC doped with other GN concentration for both unannealed status and 120°C annealing. The measured TDS spectra in unannealed P3HT/PCBM/3 wt% GN based OSC indicate one desorption species attributed as water vapor. The first desorption peak at 130°C corresponds to the glass transition temperature of PCBM, a turning point in P3HT/PCBM/3 wt% GN. Annealing at 120°C on P3HT/PCBM/3 wt% GN can casue P3HT as well as PCBM have comparatively good crystallization,

which in turn aids transport of charge carriers in P3HT/PCBM/3 wt% GN based OSC. The P3HT/PCBM based OSC with 3 wt% GN doping and 120°C annealing on photoactive layer exhibit optimized doping concentration and annealing temperature.

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