

Synergistic Optimization of Double Layer Capacitance and Pseudocapacitance of Activated Carbon by Nickel Oxide Loading

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Received: 7 July 2019 / Accepted: 18 September 2019 / Published: 29 October 2019

Low concentration basic nickel carbonate is selected to load nickel oxide on sargassum-based activated carbon to improve the gravimetric capacitance of activated carbon electrode materials. Based on the pyrolysis characteristics of basic nickel carbonate, the nickel oxide is introduced on the surface of activated carbon, while the effect of load on the original pore structure of activated carbon is reduced, which leads to the synergistic optimization of double layer capacitance and pseudocapacitance of activated carbon. In addition, the effects of nickel oxide loading at different concentrations on the pore structure and electrochemical properties of sargassum-based activated carbon are also studied. The results show that: after nickel oxide loading modification, activated carbon retains the original pore structure well, and a large amount of nanoscale nickel oxide are loaded on the surface of pore structure evenly. The modified activated carbons have both double layer capacitance and pseudocapacitance. The gravimetric capacitance performance of activated carbon has been improved obviously. Besides, with the increase of nickel oxide loading concentration, the gravimetric capacitance performance of activated carbon has been improved gradually.

Keywords: Activated carbon; Nickel oxide; Double layer capacitance; Pseudocapacitance; Synergistic optimization

1. INTRODUCTION

As a new type of green electrical energy storage devices, supercapacitors narrow the gap between conventional dielectric capacitors and batteries on account of their high energy and power densities [1-4]. Among them, carbon-based supercapacitors are the most widely used supercapacitors in the world [5-7], more than 80% of the commercial supercapacitors are fabricated based on porous carbon materials [8-10], but single carbon electrode capacitors are often limited by their low gravimetric capacitance.

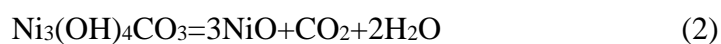
Activated carbon is the most popular electrode material for carbon-based supercapacitors [11-14], but its application in supercapacitors is limited to a large extent due to its poor conductivity, low density and small capacitance generated by the double layer principle [15,16]. Activated carbon is very suitable for negative materials of supercapacitors, but its performance is not ideal enough when it is used as positive materials [17]. Therefore, supercapacitors can be designed as asymmetric capacitors. One of the electrodes is a double layer electrode material, the other is a pseudocapacitive material. Nowadays, the common negative electrode materials for asymmetric capacitors are mainly activated carbon with huge specific surface area, while the positive electrode materials are mainly metal oxides and conductive polymers with pseudocapacitive properties [18]. Thin film electrodes made of metal oxides and conductive polymers can be used as positive electrodes in asymmetric capacitors, which exhibits good electrochemical performance in electrolyte. However, thin film electrodes show poor strength property during charging and discharging process. The active material is easily detached from the electrodes, which affects the cycle stability of asymmetric capacitors deeply and leads to a shorter life of supercapacitors. Compared with thin film electrodes, porous carbon electrodes have great advantages in chemical stability and cyclic stability in electrolyte [19,20]. Therefore, activated carbon can be compounded with metal oxides or conductive polymers, activated carbon with abundant porous structure can be used as matrix to load pseudocapacitive materials on its surface, combining the advantages of two kinds of electrode materials, so as to enhance the electrochemical properties of the electrodes. On the one hand, activated carbon has a huge specific surface area, which can generate double layer capacitance. It can also provide enough space for the adsorption of pseudocapacitive materials on the surface of activated carbon. Meanwhile, the abundant porous structure provides transport channels for electrolyte ions and charges. On the other hand, the pseudocapacitive material on the surface of activated carbon can produce a large number of pseudocapacitance, which can improve the gravimetric capacitance of the electrode material.

A lot of research has been done on composite electrode materials [21-26], but most of them only use activated carbon as the framework to provide adsorption surface for pseudocapacitive materials, and pseudocapacitive materials are usually compounded with activated carbon in a large proportion. However, the research of loading pseudocapacitive materials on the surface of activated carbon in a very small proportion has hardly occurred. Because of the small load proportion, the pore structure of activated carbon is almost unchanged while introducing pseudocapacitance, which leads to the synergistic optimization of double layer capacitance and pseudocapacitance. In this study, nickel oxide was loaded on the surface of activated carbon by pyrolysis of low concentration basic nickel carbonate (BNC). On the basis of keeping the original pore structure of activated carbon as far as possible, nickel oxide was loaded on the surface of activated carbon to make the composite electrode have both double layer capacitance and pseudocapacitance, so as to improve the electrochemical properties of asymmetric capacitor assembled by composite electrode and carbon electrode. The nickel oxide/activated carbon composite electrode was prepared, and its surface morphology and pore structure were observed. The composite electrode was used as the positive material of asymmetric capacitors, and the sargassum-based activated carbon was used as the negative electrode material to assemble asymmetric capacitors, and its electrochemical properties were studied.

2. EXPERIMENTAL

2.1 Modification mechanism

Nitrates usually have excellent water solubility and poor thermal stability, such as $\text{Ni}(\text{NO}_3)_2$, $\text{Fe}(\text{NO}_3)_3$ and $\text{Cu}(\text{NO}_3)_2$, etc.. Nitrate dissolved in water can be efficiently adsorbed by activated carbon with strong adsorbability, and decomposed at high temperature. Metal oxides produced by pyrolysis are evenly loaded on the surface of activated carbon, so as to achieve the purpose of loading modification of activated carbon. However, as nitrate enters into the pore structure of activated carbon, the metal oxides produced by pyrolysis adsorb on the pore surface, which will lead to the decrease of pore size and the blockage of small pore structure. As a result, the specific surface area of activated carbon decreases, and the ability of pore structure to transport electrolyte ions decreases, resulting in the decrease of double layer capacitance of activated carbon. In order to load metal oxides on the pore structure surface of activated carbon, and reduce the influence of metal oxides on the pore structure at the same time. BNC was selected to load nickel oxide on sargassum-based activated carbon. The pyrolysis reaction of BNC at high temperature can be divided into two steps. The first step is the dehydration reaction of BNC, and the second step is the decomposition reaction of anhydrous salt. The specific reaction formula is as follows:



Because of the strong adsorbability of activated carbon, BNC can easily enter into the pore structure of activated carbon and be adsorbed evenly at the active site. Therefore, nickel oxide produced by BNC pyrolysis has good dispersion and uniformity in activated carbon. In addition, only a small amount of BNC enters into the pore structure of activated carbon when the concentration of BNC is low, so most of the nickel oxide particles are nanometer scale. However, the activity of nickel oxide produced by pyrolysis at high temperature is higher, which easily leads to the agglomeration of nickel oxide.

It can be seen from the reaction equation (2) that the decomposition of BNC at high temperature produces nickel oxide, while carbon dioxide and water vapor are produced. These two oxidizing gases can react with carbon atoms at high temperature. Because most of these gases are produced in the pore structure of activated carbon, they can react directly with carbon atoms on the pore wall of activated carbon, which is conducive to the increase of pore size of activated carbon, thus reducing the influence of loading on the pore structure of activated carbon.

2.2 Modification experiment

188.1, 282.2, 376.2 and 752.4 mg BNC were accurately weighed and dissolved in 500 mL deionized water, respectively. Resulting in the preparation of BNC solutions with molar concentrations of 1×10^{-3} , 1.5×10^{-3} , 2×10^{-3} and 4×10^{-3} mol L^{-1} . Four portions of sargassum-based activated carbon AC₃₁₅₅ (activated carbon with a specific surface area of 3155 $\text{m}^2 \text{g}^{-1}$) [27] with a mass of one gram were obtained by weighing and put into four BNC solutions of different concentrations, respectively, and stirred for six hours in a water bath pot at 60 °C by magnetic stirring. The impregnated activated carbon

was fully dried in a drying oven at 100 °C, then heated to 900 °C at a rate of 10 °C min⁻¹ in a muffle furnace for 90 minutes in nitrogen atmosphere, and finally cooled to atmospheric temperature to obtain the modified activated carbon. The modified activated carbon was labeled as ACNi-1, ACNi-1.5, ACNi-2 and ACNi-4 according to the concentration of BNC.

2.3 Experimental techniques

Pore structure parameters of activated carbon were carried out by measuring the N₂ adsorption-desorption at 77K on an automatic apparatus (Micromeritics ASAP2020). Combined with the BET method, isotherms were employed to calculate the specific surface area at a relative pressure range of 0.05~0.3. And the BJH and HK methods were used for calculating pore volume and pore size distributions of activated carbon mesopore and micropore, respectively. The Supra 55 scanning electron microscope (SEM) (Carl Zeiss AG, Germany) was used for observing the surface morphology of the activated carbons. The samples were characterized by powder X-ray diffraction (Rigaku D/MAX-2500PC, equipped with Cu radiation, $\lambda=1.5406 \text{ \AA}$) with a step size of 0.05° from 10 to 90°. The voltage supplied was 50 kV with a current of 150 mA. The cyclic voltammetry, galvanostatic charge-discharge and AC impedance measurement were measured on an electrochemical station (CS310H, CorrTest). The voltage window and scan rate were set to be 0–1 V and 200 mV s⁻¹, respectively. The galvanostatic charge-discharge measurement was carried out at the current density ranging from 0.1–10 A g⁻¹. The test conditions of AC impedance performance of capacitors in this study are that the test frequency range is 0.001 Hz–100 kHz, and the amplitude of AC signal is 10 mV.

3. RESULTS AND DISCUSSION

3.1 Effect of nickel oxide loading on pore structure of activated carbon

The specific surface area, micropore surface area, pore volume, micropore volume, average pore size and pore size distribution of activated carbon are calculated based on the isotherm data of nitrogen adsorption-desorption of activated carbon. Table 1 shows the pore structure parameters of the five groups of activated carbon.

Table 1. Pore structure parameters of five groups of activated carbon

Sample	S_{BET} (m ² g ⁻¹)	S_{Mic} (m ² g ⁻¹)	V_{Tot} (cm ³ g ⁻¹)	V_{Mic} (cm ³ g ⁻¹)	D_{ave} (nm)
AC ₃₁₅₅	3155	2974	2.37	1.86	2.6
ACNi-1	2575	2410	1.31	1.14	2.7
ACNi-1.5	2528	2412	1.31	1.15	2.7
ACNi-2	2323	2213	1.13	1.00	2.6
ACNi-4	1128	1113	0.47	0.41	2.7

Note: S_{BET} represents specific surface area; S_{Mic} represents micropore specific surface area; V_{Tot} represents total pore volume; V_{Mic} represents micropore volume; D_{ave} represents average pore size.

It can be seen from the Table 1 that obvious changes of activated carbon pore structure parameters are occurred after nickel oxide loading. The specific surface area of activated carbon decreases from $3155 \text{ m}^2 \text{ g}^{-1}$ to 2575, 2528, 2323 and $1128 \text{ m}^2 \text{ g}^{-1}$, respectively. Except for the parameters of ACNi-1 and ACNi-1.5, the specific surface area and pore volume of activated carbon decrease significantly with the increase of BNC concentration. Especially when the molar concentration of BNC is $4 \times 10^{-3} \text{ mol L}^{-1}$, the decrease amplitude of activated carbon specific surface area and pore volume is obviously increased. When the concentration of BNC increases from $1 \times 10^{-1} \text{ mol L}^{-1}$ to $2 \times 10^{-3} \text{ mol L}^{-1}$, the specific surface area and pore volume of activated carbon decrease by 9.9% and 13.7%. However, compared with ACNi-2, the specific surface area and pore volume of ACNi-4 decreased dramatically by 51.4% and 58.4%. This is because, the amount of nickel oxide adsorbed on the surface of pore structure is increasing with the increase of BNC concentration, which leads to the decrease of pore size of activated carbon. In addition, some nanometer scale nickel oxide agglomerate and block some small pore structure, which results in the inaccessibility of nitrogen in the process of nitrogen adsorption-desorption, thus the calculated specific surface area and pore volume decrease significantly. $\text{Ni}(\text{NO}_3)_2$ is usually selected for the preparation of NiO/AC composites. It was found that the specific surface area of activated carbon decreased from $1290 \text{ m}^2 \text{ g}^{-1}$ to $250 \text{ m}^2 \text{ g}^{-1}$ after loading nickel oxide at calcining temperature of $300 \text{ }^\circ\text{C}$, and even decreased to $2 \text{ m}^2 \text{ g}^{-1}$ at calcining temperature of $800 \text{ }^\circ\text{C}$ [28]. By contrast, the specific surface area of activated carbon was well preserved by calcining BNC to prepare composite electrodes. Nickel oxide is produced during calcination of BNC, which also produces water and carbon dioxide. The load of nickel oxide will increase the pseudocapacitance of the electrode material, while the formation of water and carbon dioxide will reduce the effect of load on the pore structure and the decrease of specific surface area, thus realizing the synergistic optimization of pseudocapacitance and double layer capacitance.

In order to observe the influence of nickel oxide loading on the pore size distribution of activated carbon more intuitively, HK and BJH methods were adopted to calculate the micropore and mesopore size distribution of activated carbon, respectively, based on the nitrogen adsorption-desorption data. The results are shown in Fig. 1.

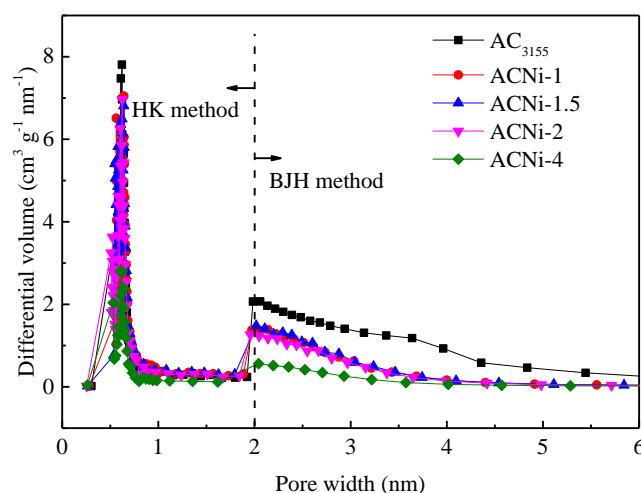


Figure 1. Pore size distribution of nickel oxide loaded activated carbon

It can be seen from the Fig. 1 that although the number of micropore and mesopore of activated carbon loaded with nickel oxide decreases in varying extents, the overall trend of pore size distribution has not changed obviously. The loaded activated carbon is still a high microporous carbon material, and the micropore size mainly distributes in 0.5-1 nm, the mesopore size mainly distributes in 2-4 nm. In addition, compared with other groups of activated carbon, the number of micropore and mesopore of ACNi-4 decreased significantly, which is consistent with the specific surface area and pore volume parameters of ACNi-4 in Table 1.

Although the specific surface area and pore number of the modified activated carbon are decreased, the specific surface area and pore volume of the modified activated carbon are still relatively large when the BNC concentration is low, and the original pore size distribution of the activated carbon is basically maintained after the loading of nickel oxide, so considerable double layer capacitance can still be generated.

3.2 Effect of nickel oxide loading on surface morphology of activated carbon

Scanning electron microscope (SEM) images of AC₃₁₅₅ and ACNi-4 are shown in Fig. 2. As can be seen from the picture, the surface of the activated carbon before modification is relatively smooth, almost no other granular substances can be observed except for some tiny activated carbon particles. A large number of nanometer scale nickel oxide particles were evenly distributed on the surface of the modified activated carbon, while some tiny particles gathered together and formed larger particles. Which indicates that nickel oxide can be loaded on the surface and pore structure of activated carbon by impregnating activated carbon with BNC solution and pyrolysis at high temperature. In addition, the SEM image of ACNi-4 shows that activated carbon still has abundant pore structure after loading a large amount of nickel oxide.

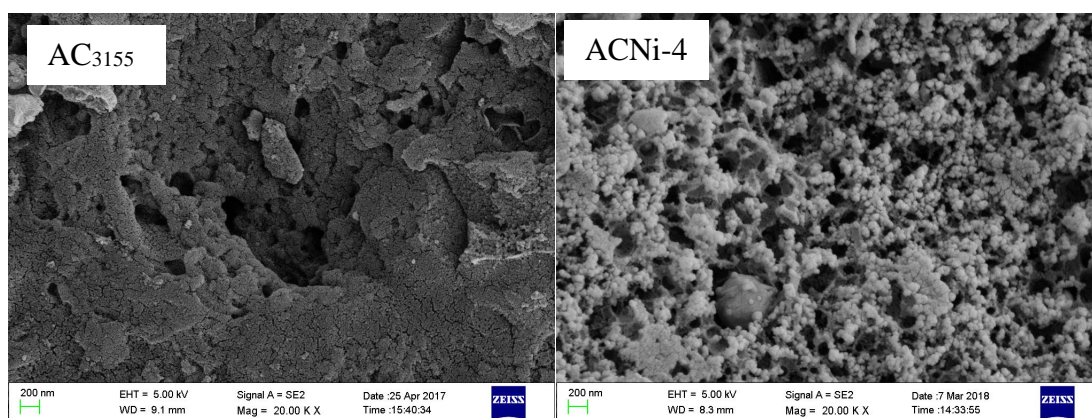


Figure 2. The surface morphology of AC₃₁₅₅ and ACNi-4

3.3 X-ray diffraction (XRD) of nickel oxide modified activated carbon

In order to further confirm whether nickel oxide is adsorbed on the surface of activated carbon and the crystal structure of nickel oxide, the modified activated carbon was tested by XRD. The XRD spectrum of ACNi-1.5 is shown in Fig. 3.

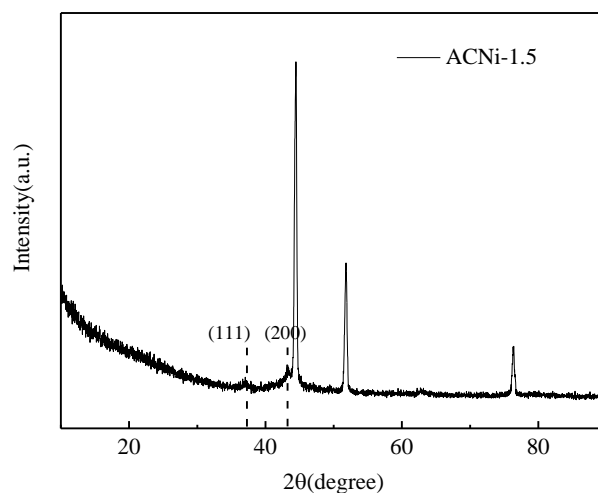


Figure 3. XRD spectrum of ACNi-1.5

It can be clearly seen that the XRD spectrum of activated carbon modified by nickel oxide show sharp characteristic peaks at the positions of $2\theta=38^\circ$ and 43° , corresponding to 111 and 200 crystal planes of nickel oxide, respectively. Which indicates that the nickel oxide is effectively adsorbed on the surface of activated carbon. Due to the low concentration of BNC and the huge specific surface area of activated carbon, the distribution of BNC on the surface of activated carbon is relatively dispersed. Therefore, nickel oxide adsorbed on the surface of activated carbon is less and exists in nanometer scale, which leads to weak characteristic peaks of nickel oxide on the modified activated carbon. In addition, from the XRD spectrum, it can be seen that nickel oxide loaded on the surface of activated carbon is hexagonal phase. Good crystallinity is conducive to the transport of electrolyte ions and charges, thus improving the electrochemical properties of composite carbon electrode materials.

3.4 Electrochemical properties of asymmetric capacitors

Activated carbon modified by nickel oxide was used as positive material and AC₃₁₅₅ was used as negative material to assemble asymmetric capacitors for electrochemical testing.

3.4.1 The mechanism of pseudocapacitance produced by nickel oxide

In pseudocapacitors, the reversible redox reaction occurs between the active material on the electrode and the electrolyte, which converts a large number of charges into high density chemical energy during charging, the corresponding capacitance is pseudocapacitance. In the process of charging

and discharging, the redox reaction is mainly accomplished by the transfer of protons and electronic defects on the semiconductor lattice. The reversible redox reaction of nickel oxide with alkaline electrolyte during charging and discharging is as follows [28]:



During the charging process, the reaction proceeds to the right, and with the transfer of electrons, the divalent nickel is oxidized to trivalent nickel, while the nickel oxide on the electrode reacts with OH^- in the electrolyte, resulting in the production of nickel oxyhydrogen, charges are also stored in supercapacitors as the oxidation reaction proceeds. The redox reaction is highly reversible during charging and discharging process, so the nickel oxide electrode usually has excellent charge-discharge efficiency. In addition, the reversible reaction proceeds very quickly in both directions, which can effectively shorten the charging and discharging time of supercapacitors. Therefore, nickel oxide is suitable as the electrode material for supercapacitors. However, there is almost no free OH^- in the electrode, and nickel oxide can only react with OH^- in the electrolyte. Therefore, the ability of the pore structure of activated carbon to transport OH^- is very important. Activated carbon provides large surface area for nickel oxide loading, while reasonable pore structure has strong ion transport capacity. It can provide enough OH^- for redox reaction, which is the key to improve the electrochemical properties of composite materials.

3.4.2 Gravimetric capacitance performance of asymmetric capacitors

The constant current charge and discharge tests are carried out at different current densities to research the gravimetric capacitance performance of asymmetric capacitors. The capacitance values are calculated according to the test results, as shown in Fig. 4. It can be seen that the gravimetric capacitance of activated carbon has been improved to a varying extent at different current densities after nickel oxide modification. When the current density is 0.1 A g^{-1} , the gravimetric capacitances of AC₃₁₅₅, ACNi-1, ACNi-1.5, ACNi-2 and ACNi-4 are 265, 295, 333, 355 and 357 F g^{-1} , respectively. Compared with the capacitance value of AC₃₁₅₅ electrode, the capacitance increased by 11.3%, 25.7%, 34.0% and 34.7%, respectively. The capacitance performance of modified activated carbon is improved greatly. With the increase of current density, the gravimetric capacitance of nickel oxide modified activated carbon decreases gradually, but the rate of reduction also decreases. Especially when the current density exceeds 3 A g^{-1} , the gravimetric capacitance of modified activated carbon tends to be flat. Which indicates that the asymmetric capacitor assembled by nickel oxide modified activated carbon and sargassum-based activated carbon has excellent high power performance. C.H. Kim et al. prepared ZnO/AC nanofiber composites by pyrolysis of zinc acetate, and obtained electrode materials with specific capacitance of 178 F g^{-1} at current density of 1 mA cm^{-2} [30]. M. Kim et al. prepared MnO_2/AC nanoneedle composites by chemical precipitation in an isopropyl alcohol/water solution system, and obtained electrode materials with specific capacitance of 271.5 F g^{-1} at a scan rate of 10 mV s^{-1} [31]. The gravimetric capacitance properties of the NiO/AC composites prepared in this study are better compared with those of many composites, and the better capacitance properties of the NiO/AC composites are mainly due to the synergistic effect of metal oxide loading and oxidizing gas pore enlargement.

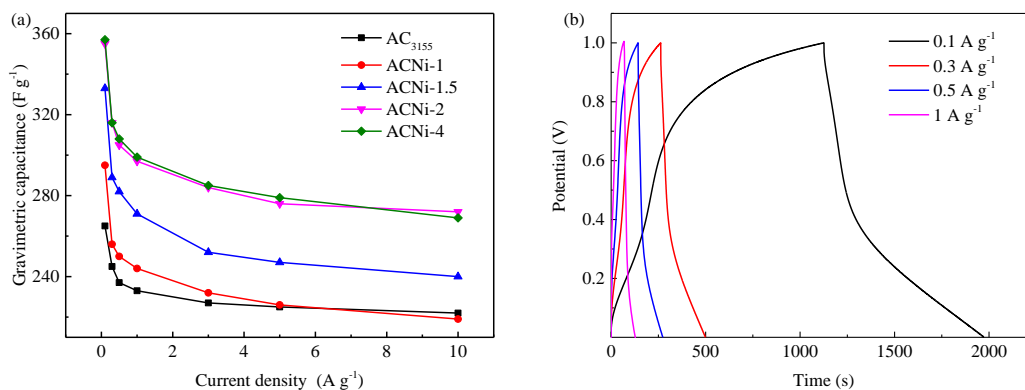


Figure 4. (a) Gravimetric capacitances of modified activated carbon at current densities of 0.1, 0.3, 0.5, 1, 3, 5 and 10 A g⁻¹; (b) Gravimetric capacitances of ACNi-4 at current densities of 0.1, 0.3, 0.5 and 1 A g⁻¹

Fig. 5 shows the relationship between the gravimetric capacitance of nickel oxide modified activated carbon and the concentration of BNC at different current densities. When the concentration of BNC increases from 1×10⁻³ mol L⁻¹ to 2×10⁻³ mol L⁻¹, the gravimetric capacitance of modified activated carbon increases with good linear characteristics, however, the gravimetric capacitance of modified activated carbon remains almost unchanged with the increase of BNC concentration from 2×10⁻³ mol L⁻¹ to 4×10⁻³ mol L⁻¹. With the increase of BNC concentration, more and more nickel oxide is loaded on the surface of activated carbon and produces more pseudocapacitance.

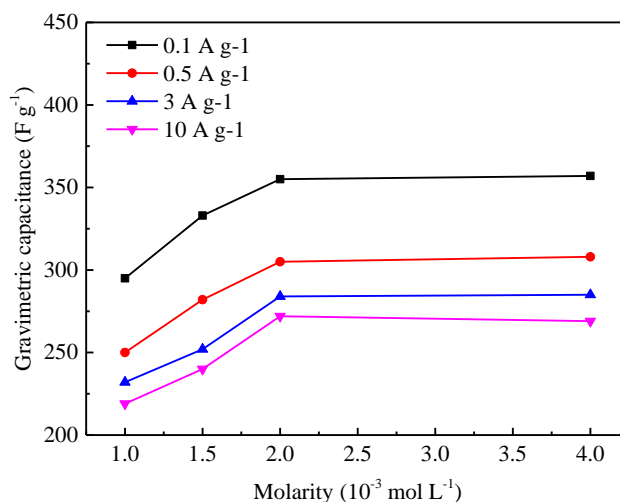


Figure 5. Relationship between gravimetric capacitance of activated carbon and concentration of BNC

At the same time, nickel oxide show little effect on the pore structure of activated carbon because of its small concentration. Carbon electrodes can still maintain good double layer capacitance performance. When the concentration of BNC exceeds 2×10⁻³ mol L⁻¹, more and more nickel oxide adsorbed on the surface of activated carbon with the increasing concentration, and some nickel oxide particles are agglomerated, which results in the blockage of some microporous structure of activated

carbon, the number of porous structure of activated carbon is significantly reduced, and the double layer capacitance of activated carbon is reduced. In addition, the capacity of pore structure to transport electrolyte ions is also reduced, and enough OH^- cannot be transported to react with nickel oxide on the surface of activated carbon, resulting in a decrease in the increase rate of pseudocapacitance. As a result, the gravimetric capacitance of modified activated carbon tends to be stable with the increase of BNC concentration.

3.4.3 Charge-discharge efficiency of asymmetric capacitors

Fig. 6 shows the relationship between charge-discharge efficiency and cycle times of asymmetric capacitors prepared by ACNi-1 and ACNi-2. As can be seen from the Fig. 6, the charge-discharge efficiency of asymmetric capacitors is relatively low at the beginning, only 72.6% and 53.8% respectively. In the process of modification, besides the introduction of nickel oxide, activated carbon will also introduce other impurities, which may react irreversibly with the electrolyte during charging and consume electric energy. In addition, the introduction and increase of some surface functional groups in the modification process may also cause irreversible reaction with electrolyte and consume electric energy. These reasons lead to low initial charge-discharge efficiency of supercapacitors. Compared with ACNi-1/AC₃₁₅₅ asymmetric capacitor, the charge-discharge efficiency of ACNi-2/AC₃₁₅₅ asymmetric capacitor is lower because more impurities and surface functional groups are introduced in the modification process with the increase of BNC concentration. With the increase of cycle times, the charge-discharge efficiency increases gradually, and after five cycles, the efficiency is almost stable at 100%. With the increase of charging and discharging times, the impurities and surface functional groups which react irreversibly with electrolyte in activated carbon are gradually consumed completely, and almost all the electric energy during charging process is stored in the form of double layer capacitance and pseudocapacitance. The nearly 100% charge-discharge efficiency of asymmetric capacitors also represents that the redox reaction between nickel oxide and alkaline electrolyte is highly reversible.

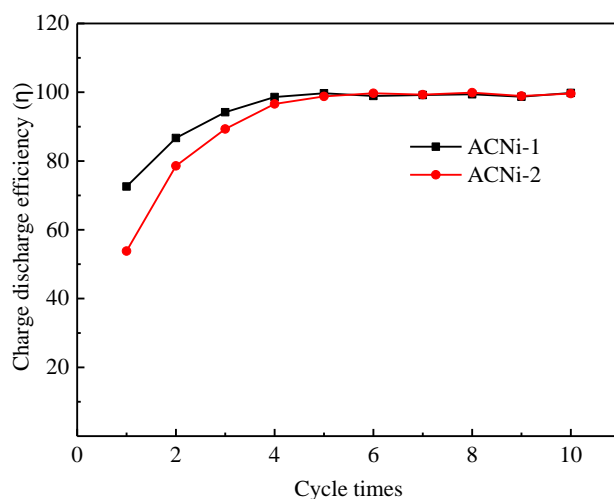


Figure 6. Relationship between cycle times and charge-discharge efficiency of asymmetric capacitors

3.4.4 Cyclic voltammetric characteristics of asymmetric capacitors

Fig. 7 shows the cyclic voltammetric curves of four groups of asymmetric electrodes at a voltage scanning rate of 200 mV s^{-1} . It can be seen from the figure that the cyclic voltammetric curve of asymmetric capacitors assembled by nickel oxide modified activated carbon and sargassum-based activated carbon is not a standard rectangle, which represents that asymmetric capacitors have both double layer capacitance and pseudocapacitance.

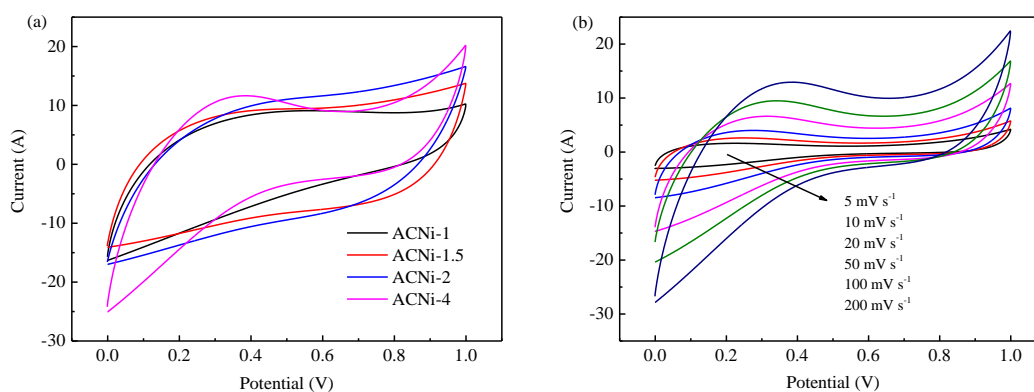


Figure 7. (a) The cyclic voltammetric curves of asymmetric capacitors at scan rate of 200 mV s^{-1} ; (b) The cyclic voltammetric curves of ACNi-4/AC asymmetric supercapacitor measured at different scan rates of 5, 10, 20, 50, 100, and 200 mV s^{-1}

3.4.5 Equivalent series resistance of asymmetric capacitors

According to the constant current charge-discharge curve of asymmetric capacitors, the equivalent series resistance of capacitors at current density of 1 A g^{-1} is calculated, the result is shown in Fig. 8. With the increase of BNC concentration, the equivalent series resistance of asymmetric capacitors increases gradually. Firstly, with the increase of the concentration of BNC, the amount of nickel oxide loaded on the surface of activated carbon increases, which leads to the increase of the resistance of the electrode. Secondly, the pseudocapacitance produced by nickel oxide in supercapacitors is realized by redox reaction, there exists charge transfer in the process of reaction, which will consume charging power and lead to the increase of supercapacitor resistance, so the resistance of supercapacitor increases with the increase of the quantity of nickel oxide. Finally, when nickel oxide is adsorbed on the surface of activated carbon, the resistance of activated carbon to ion transport of electrolyte increases, which also leads to the increase of equivalent series resistance of supercapacitors.

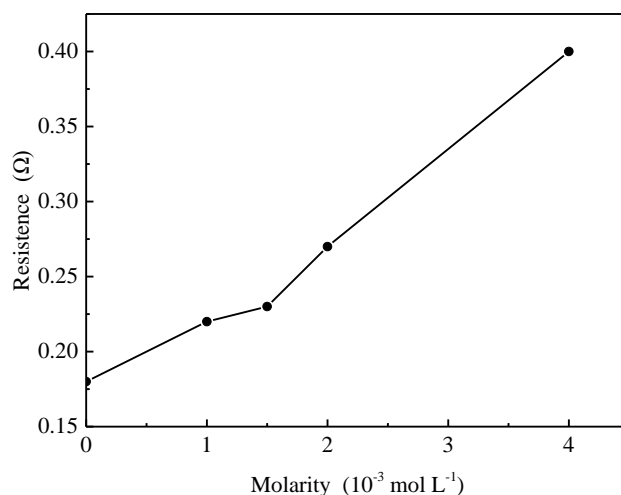


Figure 8. Equivalent series resistance of asymmetric capacitors at current density of 1 A g^{-1}

3.4.6 Cyclic performance of asymmetric capacitors

Supercapacitors are charged and discharged 10000 times at a constant current density of 5 A g^{-1} to study their cycle performance, the results are shown in Fig. 9. Compared with the symmetrical capacitor assembled by AC₃₁₅₅, the cyclic stability of the asymmetrical capacitors assembled by ACNi-2 and AC₃₁₅₅ is lower, but they still show excellent cyclic stability performance. After 10000 cycles, the gravimetric capacitance retention rate reaches 83.5%, which indicates that nickel oxide modified activated carbon still maintained good chemical stability in alkaline electrolyte. The capacitance decay of the symmetrical capacitor assembled by AC₃₁₅₅ occurs mainly in the first 3000 cycles, while the capacitance decay of the asymmetrical capacitor occurs in the whole cycle process. This is because, with the increase of cycle times, some nickel oxide adsorbed on the surface of activated carbon gradually detached from the electrode, resulting in a decrease in pseudocapacitance. In addition, with the increase of cycle times, a small amount of pore structure of activated carbon collapsed, resulting in the reduction of double layer capacitance. L.Q. Fan et al. fabricated asymmetric supercapacitor based on graphene oxide/polypyrrole composite and activated carbon electrodes, and after 5000 cycles, the capacitor keeps 85.7% of the initial available capacitance [32]. X. Du et al. researched electrochemical performances of nanoparticle Fe₃O₄/activated carbon supercapacitor, and the cycling performance that the supercapacitor can keep 82% of initial capacity over 500 cycles was observed [33]. By contrast, the NiO/AC composites prepared in this study have better cyclic stability, which is mainly attributed to the stable physical and chemical properties of activated carbon, but also shows that nickel oxide is firmly loaded on activated carbon.

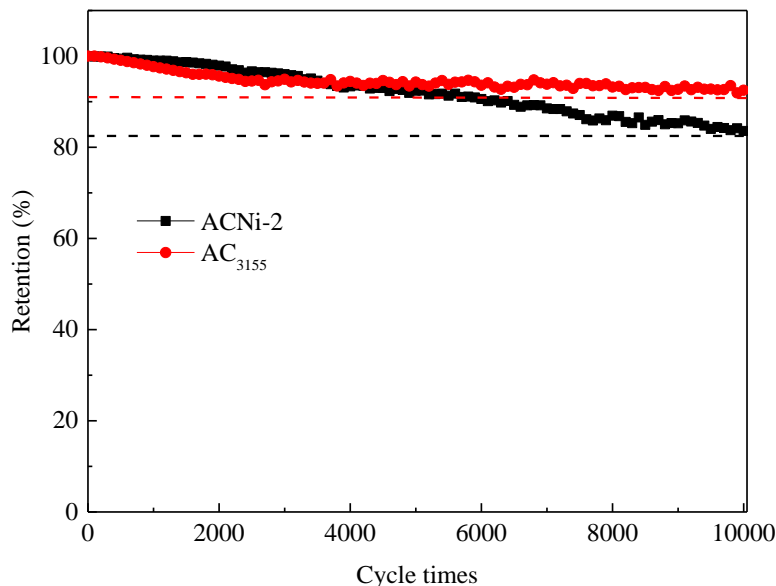


Figure 9. Cyclic performance of asymmetric capacitors at current density of 5 A g^{-1}

4. CONCLUSION

In this study, BNC with low concentration was used to impregnate activated carbon. The impregnated activated carbons were pyrolyzed at high temperature in nitrogen atmosphere to accomplish the loading of nickel oxide on the surface of activated carbon. Nickel oxide modified activated carbon and sargassum-based activated carbon were assembled into asymmetric capacitors to study their electrochemical properties.

(1) According to nitrogen adsorption-desorption, XRD and SEM tests, a large number of nanometer scale nickel oxide particles are adsorbed on the surface of activated carbon, and the modified activated carbon still have abundant pore structure.

(2) Compared with the symmetrical capacitor assembled by AC₃₁₅₅, the gravimetric capacitance of the asymmetrical capacitor assembled by nickel oxide modified activated carbon and AC₃₁₅₅ is obviously increased. With the increase of the concentration of BNC solution, the gravimetric capacitance increases gradually, but the increase rate slows down.

(3) Compared with symmetrical capacitor, the equivalent series resistance of asymmetrical capacitors assembled with modified activated carbon increases due to the poor conductivity of nickel oxide, and the resistance of capacitor increases gradually with the increase of nickel oxide content.

(4) As the nickel oxide on the surface of modified activated carbon partially detached from the electrode with the increase of cycle times, the cyclic stability of asymmetric capacitors decreased, but still show excellent cyclic stability.

ACKNOWLEDGEMENTS

The authors are grateful to the Shandong University for providing equipment support. This study was supported by the Doctoral Fund of Shandong Jianzhu University (XNBS1838), Fundamental Research Funds of Shandong University (2016JC005), Natural Science Foundation of Shandong, China (ZR2017MEE010) and Shandong Provincial Natural Science Foundation (ZR2019BEE059).

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