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# **Coloration and Ion Insertion Kinetics Study in Electrochromic** WO<sub>3</sub> Films by Chronoamperometry

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Devices relying on reversible ion intercalation are used in sustainable energy technology, for example, in batteries, in rechargeable electrical and electrochromic (EC) glazing for energy efficient buildings. Degradation of electrochromic thin films under extended charge insertion/extraction is a technologically important issue for which no in-depth understanding is currently available. The influence of coloration depth on the performance characteristics and lifetime of electrochromic WO3 materials is investigated using chronoamperometry technology in this work. It is shown that depending on the cycling time, and the applied voltage, two different types of adsorption sites, viz. the shallow sites, surrounded by low-energy barriers and deep sites with high-energy barriers are created in the WO3 host structures. During the bleaching process, higher cycling time and applied voltage create a large coloration depth leading to trapped immobile Li<sup>+</sup> ions hindering free ion transportation and subsequent degradation of electrochromic materials and provides a general guideline suggesting the use of low voltage and cycling time for controlling the charge insertion and extraction in electrochromic materials by avoiding the large coloring depths.

Keywords: WO3 film; Coloration Depth; Chronoamperometry Technology; Trapped ions

# **1. INTRODUCTION**

The electrochromic (EC) phenomenon is considered to have reversible optical properties under the application of a voltage. The reversible electrochemical reactions change the reflectance, absorbance, and transmittance of the materials. [1] Electrochromic films have received more and more attention due to their promising applications in many fields (e.g., self-modified rear-view car mirrors and sun-protective smart windows). [2-5] Compared to polymer-based materials, metal oxide electrochromic films show distinct advantages, including high thermal and mechanical stability.[6] Among these oxides, tungsten trioxide (WO<sub>3</sub>) is a promising electrochromic material and has been used in smart windows applications. [7] WO<sub>3</sub> films appear deep-blue and colorless under the applications of negative and positive electrical fields, respectively. [8] In Li<sup>+</sup> electrolyte, the electrochromic features of WO<sub>3</sub> films are related to the insertion/extraction behavior of Li<sup>+</sup> ions according to the following reaction:[9]

 $WO_3 + xLi^+ + xe^- \leftrightarrow M_xWO_3.$  (1)

where x represents the mole fraction of inserted  $Li^+$  ions in the WO<sub>3</sub> structure. Long-term durability of the electrochromic films is an essential requirement for optical modulation necessary for industrial applications. Under negative bias, the insertion of Li<sup>+</sup> ions into the WO<sub>3</sub> network results in a deep blue color response. Subsequently, under the application of a positive bias voltage, the Li<sup>+</sup> ions get extracted out, and the color of WO<sub>3</sub> films switches from dark blue to colorless. After the application of some voltage cycles, degradation in optical modulation properties of WO<sub>3</sub> films is observed, [10, 11] as the inserted Li<sup>+</sup> ions cannot be completely extracted out of WO<sub>3</sub> films. [12, 13] Upon further voltage cycling, the trapped  $Li^+$  ions show gradual accumulation in the host structure, causing decay of the electrochromic performance. Granqvist et al. have reported two types of trap sites in WO<sub>3</sub> host structures, include: "shallow trap sites" surrounded by low energy barrier and "deep trap sites" with high energy barrier. [14] When the Li<sup>+</sup> ions diffuse into the shallow trap sites, the ions can be completely extracted out from the structure under the application of a reverse electric field. However, when the ions are driven into the deep trap sites, the inserted Li<sup>+</sup> ions will be trapped and become immobile due to their high energy barrier. This contributes to the generation of irreversible Li<sup>+</sup> ions in WO<sub>3</sub> films. Although the trapped ions could be eliminated by constant-current-drive [15, 16], the formation mechanisms of trapped ions are still unclear. Some reports attribute the formation of trapped ions to the effects of the operating voltage range. [13, 17, 18] However, this is inclusive and the effects of trapped ions on the optical modulation ability and cycling lifetime of electrochromic WO<sub>3</sub> films are still not clearly understood and need further attention.

As is well known, the charge/discharge depth of electrode materials in ion battery systems is related to its capacity fade, coulombic efficiency, and cycle life [19-21]. Especially, over-charge/discharge result in permanent internal damage, and irreversible changes in lifetime [22, 23]. Also, the influences of charge/discharge depth on the performance and lifetime of electrochromic materials have not yet been investigated. The coloration of WO<sub>3</sub> is attributed to the simultaneous insertion of Li<sup>+</sup> ions and electrons into the WO<sub>3</sub> host structure. The inserted ions into the film take part in the coloration reaction. Thus the charge depth of electrode is related to the coloration depth. However, to the best of our knowledge, the effects of coloration depth and Li<sup>+</sup> ion insertion on the performance of WO<sub>3</sub> films have not yet been systematically studied.

Chronoamperometry is an efficient tool to explore the ion diffusion process in active electrodes.[24, 25] In this study, the effects of coloration depth on the trapped ion formation and degradation of WO<sub>3</sub> film are explored via chronoamperometry technology with different duration cycling times of the potential in the coloration process. We have observed that the coloration depth has a crucial role on the electrochromic performance of WO<sub>3</sub> films, and especially, the number of trapped ions in WO<sub>3</sub> increases with the deepening of the coloring state resulting in a severe degradation after some cycles. The degradation mechanism is attributed to the over coloration depth resulting from a

long-time duration of potential. The inserted  $Li^+$  ions have enough energy to diffuse into 'deep' sites surrounded by a high-energy barrier in the WO<sub>3</sub> host structure, resulting in immobile ions.

### 2. EXPERIMENTAL SECTION

#### 2.1 Preparation of WO<sub>3</sub> thin film

The WO<sub>3</sub> thin film electrodes were prepared by depositing WO<sub>3</sub> thin films on the glass substrates pre-coated with ITO (indium tin oxide thin film) having a sheet resistivity of ~17 ohm/sq. Before deposition, the ITO coated glass substrates were cleaned ultrasonically in deionized, alcohol, and then dried in the air atmosphere, respectively. Then, the WO<sub>3</sub> films were deposited via a direct current magnetron sputtering system, in which a background pressure was set at  $2.0 \times 10^{-3}$  Pa. A metallic W target of 100 mm diameter was used. The distance between the target and substrate holder was about 7 cm, with an incident angle of 41°. Besides, the gas-flow rate of O<sub>2</sub>/Ar was fixed at 9 sccm/27 sccm. During sputtering, the power on the target was set at 280 W, and the total pressure was maintained at about 2.2 Pa. All the films were prepared with a sputtering time of 20 mins at room temperature.

### 2.2 Electrochemical and optical measurements

The electrochemical performances of WO<sub>3</sub> electrodes were measured in a three-electrode system, in which 1 M LiClO<sub>4</sub> electrolyte prepared by dissolving lithium perchlorate into propylene carbonate. The WO<sub>3</sub> film electrode was used as the working electrode. A platinum sheet and an Ag/AgCl electrode were used as the counter and the reference electrodes, respectively. For Chronoamperometry measurements, the applied voltage was set to +1 V to -1 V for different durations of 4 s, 8 s, 13 s, and 18 s, respectively, to explore the influence of coloration depth on the cycle life of WO<sub>3</sub> film and trapped ions. The electrochemical measurements were carried out using electrochemical workstation (Model Princeton VersaSTAT 4). The transmittance of WO<sub>3</sub> electrodes at 550 nm were observed via ultraviolet-visible–near-infrared spectrophotometer (Model Shimadzu UV-3101PC) during the electrochemical cycling tests.

#### **3. RESULTS AND DISCUSSION**

Chronoamperometry technology has been employed to estimate the response time and ion capacity of the electrochromic films [26-31]. In previous works, chronoamperometry technology has been applied to explore the electrochromic mechanism of WO<sub>3</sub> and NiO films, demonstrating an extraordinary role of chronocoulometry in estimating the electrochromic performance and explaining the corresponding mechanisms [32]. Besides, the chronocoulometry technology has a distinct advantage of synchronously monitoring the ion charge density in the reaction process, compared to traditional cyclic voltammetry (CV). Moreover, chronoamperometry is a useful tool to investigate ion diffusion dynamics in electrodes [33, 34]. However, to the best of our knowledge, the coloration depth has not been comprehensively investigated for controlling the coloration potential signature in chronocoulometry technology. It is well known that the insertion of Li+ ions into the host structure

during the charging process results in the coloration process. However, during the discharging process, the inserted ions get extracted out of the film, leading to a bleaching response. In this study, the coloration depth of WO<sub>3</sub> thin film is controlled by setting up different duration cycling time of the potential during the ions insertion process. In the three-electrode test system, the constant voltage discharge and charge processes can be achieved by employing double-potential step chronoamperometry technique. The response current as a function of time is recorded, and the current-time relationship is given by the Cottrell equation[35]:

$$i = nFAC_0(D/\pi t)^{1/2}$$
 (2)

where A represents the electrode area, F represents the Faraday constant, and  $C_0$  and D represent Li<sup>+</sup> ion concentration in electrolyte and Li<sup>+</sup> ion diffusion coefficient, respectively.



Figure 1. (a) Chronoamperometric curves of different potential duration time. (b) The charge-time curves during the chronoamperometric test. (c) The charge density of trapped ions in the WO<sub>3</sub> host structure under different test conditions. (d) In-situ transmittance-time relationship at  $\lambda$ =550 nm.

Figure 1(a) shows the current-time curves of the WO<sub>3</sub> electrode under the double-step potential of -1 V to +1 V versus SCE (aturated silver chloride electrode) for different time steps of 4 s, 8 s, 13 s, and 18 s. For all the samples, the current density in the bleaching process could reach to zero while the coloration current density shows a constant value after the step time. It is because a large quantity of WO<sub>3</sub> can react with the Li<sup>+</sup> ions in the coloration process to produce Li<sub>x</sub>WO<sub>3</sub>, and eventually, the coloration reaction reaches a balanced state due to sufficient supply of ions in the WO<sub>3</sub> sites. In the bleaching process, the resultant Li<sub>x</sub>WO<sub>3</sub> decompose to generate WO<sub>3</sub>, and then the current density

reaches to zero value when the generated  $\text{Li}_x \text{WO}_3$  is completely decomposed,[36]. Besides, the current density during the bleaching process is larger than that of the coloration process, and the bleaching current shows a rapid decay. The coloration response time ( $t_c$ ) and bleaching response time ( $t_b$ ) are obtained by the current-time data as shown in Fig. 1(a). The bleaching response time is found to be faster when compared to the coloring process due to the higher conductivity of the tungsten bronze conductor ( $\text{Li}_x \text{WO}_3$ ) than that of insulator WO<sub>3</sub>.[37]



**Figure 2**. The evolutions of Chronoamperometry curves of WO<sub>3</sub> film with a fixed coloring potential -1 V for different duration time of (a) 4 s, (b) 8 s, (c) 13 s, and (d) 18 s.

The charge density of  $Li^+$  ions involved in the coloring reaction is calculated by the corresponding current-time curve and can be expressed with the following equation:

$$Q = \int_0^{\tau} I \, \mathrm{dt} = 2nFAD^{1/2}C_0 \pi^{-1/2} t^{1/2} \tag{3}$$

where "Q" represents the number of ions. As shown in Fig. 1b, the long-time potential duration will contribute more ions in the electrochromic reaction. Specifically, the inserted ions are 19.15 mC cm<sup>-2</sup>, 36.38 mC cm<sup>-2</sup>, 43.96 mC cm<sup>-2</sup> and 51.19 mC cm<sup>-2</sup> in the films with -1 V versus SCE for 4 s, 8 s, 13 s and 18 s, respectively. Besides, the charge densities do not return to zero during the ion extraction process, suggesting that a part of inserted Li<sup>+</sup> ions have not extracted out of the WO<sub>3</sub> film. These Li<sup>+</sup> ions are defined as trapped ions, and the relationship between the quantity of trapped

 $Li^+$  ions and duration time of potential is shown in Fig. 1(c), which suggests that the long-time application of the electrical field will result in the formation of more trapped  $Li^+$  ions in the film. Usually, the quantity of inserted  $Li^+$  ions determines the optical modulation property of electrochromic film films.



**Figure 3.** Evolution of the charge density of inserted and extracted Li<sup>+</sup> ions in WO<sub>3</sub> film under different coloring duration time of (a) 4 s, (b) 8 s, (c) 13 s, and (d) 18 s.

Figure 1(d) shows the recorded transmittance-time curves of WO<sub>3</sub> films. The film films under the longest potential duration time of 18 s shows the lowest coloration depth of about 1.5% in optical transmittance and the largest optical modulation ability of about 89.7% as calculated from the difference between the bleaching transmittances and coloration transmittances. In the same way, the coloration transmittance of the WO<sub>3</sub> film with 4 s, 8 s, and 13 s is 16.9%, 7.4%, and 2.9%, respectively. These results suggest that the WO<sub>3</sub> films show various ion storage ability, optical modulation ability, and different quantity of trapped ions under the different coloration depths.



**Figure 4.** Cyclic evolution of (a) reversibility and (b) trapped ion density of the WO<sub>3</sub> films under different coloring time of 4 s, 8 s, 13 s, and 18 s.

The performance of the electrochromic WO<sub>3</sub> films usually suffer degradation mentioned above when the trapped Li<sup>+</sup> ions accumulate in the host structure. The above results demonstrate that the formation of trapped ions in WO<sub>3</sub> films is affected by the coloration depth. It is necessary to study the multi-cyclic performance of the electrode under the different coloration depth to analyze the decay mechanism. As shown in Fig. 2, the WO<sub>3</sub> films exhibit different cyclic stabilities under different coloration depth. Mainly, the WO<sub>3</sub> film under a coloration time of 4 s exhibits the highest cyclic stability (Fig. 2a). The current density and peak shape show inconspicuous change during 3000 cycles. When the duration time of the potential increase to 8 s, the stability of the WO<sub>3</sub> film shows some decrease (Fig. 2b). Further, the film shows obvious degradation when the duration time of the potential increase to 18 s (Fig. 2d). The current density also decreases to almost zero value after 3000 cycles, indicating a week electrochromic activity of the WO<sub>3</sub> electrode.

The electrochromic stability could further be explored by the corresponding cycling evolution of ion charge density calculated by equation (3). As expected, the relationship between duration time of potential and stability of charge capacity is consistent with the current density evolution. Mainly, for the step potential of -1 V for 4 s, the charge density exhibits the highest stability, and , not much decay could be observed, as shown in Fig. 3(a). When the duration time of the potential increases to 8 s, the charge capacity shows some degradation. Further, this trend is magnified by increasing the duration time to 18 s (Fig. 3d). The film shows a larger ion capacity at the initial few cycles, then gradually decreases and finally gets close to zero upon further cycling. These results indicate that the coloration depth significantly affect the ion storage capacity of WO<sub>3</sub> films. Also, the inserted charge density shows some difference with the extracted charge density, suggesting the irreversible accumulation of inserted Li<sup>+</sup> ions in the film.

Usually, the electrochromic films with high stability should prohibit the accumulation of irreversible ions in the films. As mentioned above, the formation of the trapped ions due to inserted ions, are difficult to extract out of the films in the subsequent bleaching process. The amount of the trapped ions ( $Q_{trap}$ ) can be calculated as follows:



**Figure 5.** (a) Multi-cyclic transmittance-time curves of WO<sub>3</sub> film at  $\lambda$ =550 nm under a fixed coloring potential -1 V for different duration time of (a) 4 s, (b) 8 s, (c) 13 s and (d) 18 s.

$$Q_{\rm ui} = \int_{1}^{m} \{ (1 - R) \times Q_{\rm in} \} \, \mathrm{d}n \tag{4}$$

where *m* is the cycle number,  $Q_{ui}$  is the accumulated quantity of "trapped ion" at the *m*-th cycle. The reversibility *R* is estimated by the ratio of extracted charge ( $Q_{ex}$ ) to inserted charge ( $Q_{in}$ ) during the reaction process[38]:

$$R = \frac{Q_{\rm ex}}{Q_{\rm in}} \times 100\%$$
 (5)

The  $Q_{trap}$  is determined from the amount of inserted ions  $Q_{in}$  and the reversibility *R*. In Fig 4 (a), the R of WO<sub>3</sub> film under the long potential duration time is lower than that of the sample with short potential duration time at the same cycle, and the corresponding  $Q_{in}$  shows a larger value. Consequently, the amount of "trapped Li<sup>+</sup> ions" increases with longer duration time of potential, as shown in Fig. 4 (b).

 $Li^+$  ion insertion in WO<sub>3</sub> films will result in a color change of the film from colorless to deepblue. In the bleaching process,  $Li^+$  ions will get extracted from the WO<sub>3</sub> host structure. The evolution of transmittance-time curves of the film is recorded during cycling, as shown in Fig. 5. Downward optical transmittance curves and negative current responses are obtained when a negative potential is applied to the film electrode during the coloration process. Upward optical transmittance curves, as well as positive current curves, are obtained during the bleaching process when positive potential is applied to the film. As shown in Figs. 5(a-b), the coloration state shows no obvious degradation, and the bleaching state remains stable during all the cycles, suggest excellent stability in optical modulation. However, under the long potential times, of 13 s and 18 s as shown in Figs. 5(c-d), the transmittance of WO<sub>3</sub> film at bleaching state shows a significant decay than the coloration state.



**Figure 6.** (a) Transmittance cyclic evolution of the film at coloration state and (b) calculated Li<sup>+</sup> ions diffusion coefficient change of WO<sub>3</sub> upon cycling under different test conditions.

The cyclic evolution of the optical transmittance of WO<sub>3</sub> films is shown in Fig. 6(a). These results suggest that the degradation of WO<sub>3</sub> electrochromic performance derives from the decay of bleaching state under long potential durability for 13 s and 18 s. As discussed above, the coloration process of WO<sub>3</sub> films corresponds to the Li<sup>+</sup> ion insertion process: WO<sub>3</sub> +  $x(\text{Li}^+ + e^-) \rightarrow \text{Li}_x \text{WO}_3$ , and the bleaching process corresponds to Li<sub>x</sub>WO<sub>3</sub>  $\rightarrow$  WO<sub>3</sub> +  $x(\text{Li}^+ + e^-)$ . The formation of Li<sub>x</sub>WO<sub>3</sub> causes a decrease in transmittance. Therefore, the transmittance decay of the films during the bleaching state cycling for 13 s and 18 s samples should be attributed to the incomplete decomposition of Li<sub>x</sub>WO<sub>3</sub> during the extraction process. The incomplete decomposition of Li<sub>x</sub>WO<sub>3</sub> is due to the formation of "trapped ions" which is consistent with the earlier observations (see Fig. 4 b). Based on these results, the stability of WO<sub>3</sub> electrodes and the formation of trapped ions show a direct relationship to the coloration depth. To explore further on the degradation mechanisms, the diffusion coefficients of Li<sup>+</sup> ions are calculated using the equation (3) and are shown in Fig. 6 b. The films under a long potential durations of 13 s and 18 s show a large decay in ion diffusion after some cycles, and Li<sup>+</sup> ions can not diffuse into the film, suggesting that the Li<sup>+</sup> ions may be driven into the isolated 'deep' trap sites due to the high-energy under a long-time potential duration.



**Figure 7.** Simple schematic diagram of the formation mechanism of trapped ions in the WO<sub>3</sub> host structure with large coloration depth.

A schematic diagram of the formation of trapped ions in the WO<sub>3</sub> host structure with considerable coloration depth is shown in Fig. 7. As mentioned above, the WO<sub>3</sub> host structure has two types of Li<sup>+</sup> ions adsorption sites: (1) shallow reaction sites surrounded by low-energy barriers which allow ions to reversibly transport throughout the film; and (2) deep sites with high-energy barriers, in which ions become immobile once trapped. The large coloration depth obtained from the long-time potential duration in the ion insertion process provide enough energy for the Li<sup>+</sup> ions to reach the ion trapped deep sites with a high energy barrier, resulting in the formation of immobile trapped ions in WO<sub>3</sub> films (Figure 7). During the ion extraction process, trapped Li<sup>+</sup> ions could not be extracted out of the WO<sub>3</sub> films accumulate in the film. No doubt, these trapped ions will further hinder ion transportation, reduce the charge capacity and optical modulation ability of the WO<sub>3</sub> films. Thus, it is better to avoid the high-energy ion diffusion process in inorganic electrochromic materials.

#### **4. CONCLUSION**

In this study, the degradation mechanisms of electrochromic WO<sub>3</sub> films were explored from different coloration depths measured using double-step chronoamperometry technology combined with an in-situ spectroelectrochemical measurement setup. It is observed that the long potential duration time will cause a considerable coloration depth and the formation of trapped ions in the WO<sub>3</sub> host structure, being the main reason for the degradation of electrochromic performance. Due to Li<sup>+</sup> ion diffusion, two different types of adsorption sites are generated: the shallow sites and deep sites, surrounded by low-energy and high-energy barriers, respectively, resulting in different coloration depths. Especially, large coloration depths due to long potential duration time will supply enough energy to drive the Li<sup>+</sup> ions into the 'deep' sites with high-energy barriers, resulting in the formation of immobile trapped ions. Further, the formation of trapped Li<sup>+</sup> ions in the WO<sub>3</sub> host structures lead to decay of the bleaching state and trapped ions will hinder free ion transportation. As a result, the

quantity of inserted  $Li^+$  ions into WO<sub>3</sub> films will decay upon cycling, resulting in the reduction of the coloration component ( $Li_xWO_3$ ). Finally, electrochromic performance of WO<sub>3</sub> films degrade. This study leads to a new understanding of the degradation mechanisms of the electrochromic materials under different coloration depths and suggests one to avoid high-energy ions diffusion process in

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inorganic electrochromic materials by reducing the potential duration time.

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