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# Enhanced Electrochemical Performance of Si-doped LiMn<sub>2</sub>O<sub>4</sub> Cathode Material for LiBs Prepared using Mn<sub>3</sub>O<sub>4</sub> Octahedrons

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We proposed a co-modification strategy of Si-doping and octahedral morphology to improve the electrochemical performance of LiMn<sub>2</sub>O<sub>4</sub>. The Si-doped LiMn<sub>2</sub>O<sub>4</sub> sample (LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons) was prepared by high-temperature solid-state method with Mn<sub>3</sub>O<sub>4</sub> octahedrons as manganese precursor and SiO<sub>2</sub> nanoparticles as silicon dopant. XRD and SEM characterization results indicated that the introduction of Si<sup>4+</sup> ions does not produce the substantive impact on the inherent spinel structure of LiMn<sub>2</sub>O<sub>4</sub> and LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons present relatively uniform particle size distribution. When cycled at 1.0 C, LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons exhibited higher initial reversible capacity than that of the undoped LiMn<sub>2</sub>O<sub>4</sub>. After 100 cycles, LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons showed better cycling stability with higher capacity retention rate of 94.7%. Moreover, LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons presented good rate capability and high-temperature cycling performance. Such good electrochemical performance has much to do with the synergistic modification of Si-doping and octahedral morphology.

**Keywords:** LiMn<sub>2</sub>O<sub>4</sub>; Silicon doping; Octahedral morphology; Mn<sub>3</sub>O<sub>4</sub> octahedrons; Electrochemical performance

# **1. INTRODUCTION**

Lithium manganate oxide (LiMn<sub>2</sub>O<sub>4</sub>) is very important cathode material for lithium-ion batteries (LiBs). Compared with other commercial cathode materials, LiMn<sub>2</sub>O<sub>4</sub> possesses several advantages such as high voltage platform, abundant manganese resource, and low production cost, which promote the widespread use of LiMn<sub>2</sub>O<sub>4</sub> in the field of energy storage and low-speed electric vehicles [1-6]. However, the poor cycling performance of LiMn<sub>2</sub>O<sub>4</sub> seriously affects the large-scale application of this cathode material to some extent.

To enhance the cycling performance, many scientific researchers optimized the electrochemical performance of LiMn<sub>2</sub>O<sub>4</sub> by using several modification strategies such as doping modification [7-15],

surface coating, controlling nanoparticles granularity, and so on. Among these strategies, the doping modification performs an important role inenhancing the electrochemical performance of LiMn<sub>2</sub>O<sub>4</sub>. The doping ions mainly involve the low valent cations ( $Cu^{2+}$ ,  $Mg^{2+}$ ,  $Al^{3+}$ )[3, 11, 16], high valent cations ( $Si^{4+}$ , Ti<sup>4+</sup>)[17-19]. Among them, the high valent cations mainly replace the tetravalent manganese ions, which can avoid the decrease of reversible capacity [5, 18]. It has been reported that the Si-doping strategy can contribute to the increasement of reversible capacity and improvement of cycling stability [17, 20]. Iturrondobeitia reported the preparation of Si-doped LiMn<sub>2</sub>O<sub>4</sub> by freeze-drying method [17]. Compared with the undoped spinel, the Si-doped LiMn<sub>2</sub>O<sub>4</sub> sample presented a higher initial discharge capacity with better cycling stability. Zhao prepared the LiSi<sub>x</sub>Mn<sub>2-x</sub>O<sub>4</sub> samples by using electrolytic manganese dioxide (EMD) as manganese precursor and tetraethylorthosilicate as silicon dopant [20]. These research works indicate that a certain amount of silicon ions can effectively enhance the electrochemical performance of LiMn<sub>2</sub>O<sub>4</sub>.

In this work, the  $LiSi_{0.05}Mn_{1.95}O_4$  octahedrons were prepared by high-temperature solid-state method with  $Mn_3O_4$  octahedrons as manganese precursor and  $SiO_2$  nanoparticles as silicon dopant. Compared with the undoped  $LiMn_2O_4$ , the introduction of  $Si^{4+}$  ions does not produce the substantive impact on the inherent spinel structure of  $LiMn_2O_4$ . Electrochemical testing shows that the  $LiSi_{0.05}Mn_{1.95}O_4$  octahedrons showed good electrochemical performance.

#### 2. EXPERIMENTAL

To prepare the Si-doped LiMn<sub>2</sub>O<sub>4</sub> sample, Mn<sub>3</sub>O<sub>4</sub> octahedrons and SiO<sub>2</sub> nanoparticles were applied as manganese precursor and silicon dopant. The LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons were prepared by high-temperature solid-state method. First of all, Mn<sub>3</sub>O<sub>4</sub> octahedrons were prepared by hydrothermal method according to the previous work [21]. A certain amount of potassium permanganate (KMnO<sub>4</sub>, 4.5 g) and ethylene glycol (C<sub>2</sub>H<sub>6</sub>O<sub>2</sub>, 7.5 ml) were dissolved successively in deionized water (100 ml). The obtained suspension was sealed in 150 ml Teflon-lined stainless steel autoclave and then maintained at 150 °C for 20 h. Based on the hydrothermal reaction process, Mn<sub>3</sub>O<sub>4</sub> octahedrons was obtained by suction filtration and dried at 80 °C for 15 h. And then, the stoichiometric mixture of LiOH·H<sub>2</sub>O, SiO<sub>2</sub> nanoparticles, and Mn<sub>3</sub>O<sub>4</sub> octahedrons were thoroughly mixed with the help of ethanol. Finally, the obtained mixture was sintered at 780 °C for 10 h in muffle furnace. By contrast, both the undoped LiMn<sub>2</sub>O<sub>4</sub> and LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> particles were prepared through the similar solid state reaction process with electrolytic MnO<sub>2</sub> as manganese precursor.

The microstructures and surface morphologies of the undoped LiMn<sub>2</sub>O<sub>4</sub> particles, LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> particles, and LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons were characterized by X-ray diffraction (XRD) and scanning electron microscope (SEM). XRD pattern was used to confirm the influence of Sidoping on the microstructure of LiMn<sub>2</sub>O<sub>4</sub>, and SEM image was applied to study the influence of Sidoping on the surface morphology of LiMn<sub>2</sub>O<sub>4</sub>. The electrochemical performance of the undoped LiMn<sub>2</sub>O<sub>4</sub> particles, LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> particles, and LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons were tested by LANHE CT2001A battery tester.

# **3. RESULTS AND DISCUSSION**



**Figure 1.** XRD patterns of (a) LiMn<sub>2</sub>O<sub>4</sub> particles, (b) LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> particles, and (c) LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons.

Figure 1 shows the XRD patterns of the undoped  $LiMn_2O_4$  particles,  $LiSi_{0.05}Mn_{1.95}O_4$  particles, and  $LiSi_{0.05}Mn_{1.95}O_4$  octahedrons. As shown here, both  $LiMn_2O_4$  particles and  $LiSi_{0.05}Mn_{1.95}O_4$  particles present obvious characteristic diffraction peaks of  $LiMn_2O_4$  (JCPDS No. 35-0782) [12, 22].



Figure 2. SEM images of (a, b)  $LiMn_2O_4$  particles, (c, d)  $LiSi_{0.05}Mn_{1.95}O_4$  particles, and (e, f)  $LiSi_{0.05}Mn_{1.95}O_4$  octahedrons.

No other impurity peaks of  $MnO_2$  or  $SiO_2$  can be observed in the above XRD patterns, which suggests the successful transformation of electrolytic  $MnO_2$  to  $LiMn_2O_4$  and integration of  $Si^{4+}$  ions in the crystal structure of  $LiMn_2O_4$  [20]. For the  $LiSi_{0.05}Mn_{1.95}O_4$  octahedrons, the corresponding XRD

pattern shows similar characteristic diffraction peaks of LiMn<sub>2</sub>O<sub>4</sub> without any other impurity peaks of  $Mn_3O_4$  and SiO<sub>2</sub>, which indicates that  $Mn_3O_4$  octahedrons were completely converted to the LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> sample [21, 23]. Furthermore, there is no (220) peak in the above XRD patterns of LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> particles and LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons. It has been reported that the emergence of (220) peak means the colonization of doped cations at tetrahedral (8a) sites [13, 24]. Thus, it can be inferred that Si<sup>4+</sup> ions only replace Mn<sup>4+</sup> ions at octahedral (16d) sites, which can enhance the spinel structural stability.

Figure 2 shows the SEM images of the undoped LiMn<sub>2</sub>O<sub>4</sub> particles, LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> particles, and LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons. It can be seen from Figure 2(a, b) that the undoped LiMn<sub>2</sub>O<sub>4</sub> particles show uneven particle size distribution. Some large agglomerated particles can be observed in the above SEM images. By contrast, the LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> particles shown in Figure 2(c, d) present relatively uniform particle size distribution, which has much to do with the introduction of Si<sup>4+</sup> ions [17, 20]. The introduction of Si<sup>4+</sup> ions contributes to the inhibition of large agglomerated particles, and the Si-doped LiMn<sub>2</sub>O<sub>4</sub> sample can present uniform particle size distribution compared to the undoped LiMn<sub>2</sub>O<sub>4</sub> particles. It should be noted that the LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> particles shown in Figure 2(c, d) also possess octahedral morphology. However, the corresponding particle size range is larger than that of LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons shown in Figure 2(e, f). For the LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons, the particle size distribution is completely different from that of the undoped LiMn<sub>2</sub>O<sub>4</sub> particles and LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons present uniform particle size distribution is completely different from that of the undoped LiMn<sub>2</sub>O<sub>4</sub> particles and LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons present uniform particle size distribution is completely different from that of the undoped LiMn<sub>2</sub>O<sub>4</sub> particles and LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons present uniform particle size distribution with smaller particle size, which may enhance the electrochemical performance of LiMn<sub>2</sub>O<sub>4</sub>.



Figure 3. Initial charge-discharge curves of the undoped  $LiMn_2O_4$  particles,  $LiSi_{0.05}Mn_{1.95}O_4$  particles, and  $LiSi_{0.05}Mn_{1.95}O_4$  octahedrons.

Figure 3 shows the initial charge-discharge curves of the undoped LiMn<sub>2</sub>O<sub>4</sub> particles, LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> particles, and LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons. These three samples were cycled at 0.5 C under the room temperature environment. It can be clearly observed that there are two obvious voltage platforms in the initial discharge curves of the undoped LiMn<sub>2</sub>O<sub>4</sub> particles. For the LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons, the corresponding voltage platforms coincide quite well with that of the undoped LiMn<sub>2</sub>O<sub>4</sub> particles, which suggests that the introduction of Si<sup>4+</sup> ions does not change the lithium intercalation process of electrochemical cycling [17, 20].



Figure 4. Cycling performance of the undoped  $LiMn_2O_4$  particles,  $LiSi_{0.05}Mn_{1.95}O_4$  particles,  $LiSi_{0.05}Mn_{1.95}O_4$  octahedrons at 0.5 C.



**Figure 5.** Histogram comparsion of initial discharge capacity and capacity retention of the undoped LiMn<sub>2</sub>O<sub>4</sub> particles, LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> particles, LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons at 0.5 C.

Figure 4 presents the cycling performance of the undoped LiMn<sub>2</sub>O<sub>4</sub> particles, LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> particles, LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons at 0.5 C. It can be seen that the undoped LiMn<sub>2</sub>O<sub>4</sub> particles exhibit poor cycling stability with low capacity retention rate of 78.4%. For the Si-doped LiMn<sub>2</sub>O<sub>4</sub> samples, both LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> particles and LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons show relatively better cycling stability (91.7% and 95.9%) compared to the undoped LiMn<sub>2</sub>O<sub>4</sub> particles. Especially, the LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons exhibit the best cycling stability with highest initial discharge capacity of 123.2 mAh g<sup>-1</sup> and capacity retention. The change trend of initial discharge capacity and capacity retention of these three samples can be intuitively observed by Figure 5. Compared with the undoped LiMn<sub>2</sub>O<sub>4</sub> particles, the good electrochemical performance is intimately connected to the co-modification of Si-doping and octahedral morphology of LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons with uniform particle size distribution [17, 23, 25].



**Figure 6.** Long cycling performance of the undoped LiMn<sub>2</sub>O<sub>4</sub> particles and LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons at 1.0 C.

Figure 6 shows the long cycling performance of the undoped LiMn<sub>2</sub>O<sub>4</sub> particles and LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons at 1.0 C. The undoped LiMn<sub>2</sub>O<sub>4</sub> particles show an initial discharge capacity of 113.1 mAh g<sup>-1</sup>. With the increasing of cycling number, the discharge capacity presents a severe attenuation with relatively low retention rate of 75.3% after 100 cycles. For the Si-doped LiMn<sub>2</sub>O<sub>4</sub> sample, LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons show relatively good electrochemical performance due to the co-modification of Si-doping and octahedral morphology [17, 21, 23]. The initial discharge capacity of LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons reach up to 118.7 mAh g<sup>-1</sup>. After 100 cycles, the corresponding capacity retention increases to 94.7%. Such good electrochemical performance has much to do with the synergistic effect of Si-doping and octahedral morphology. Table 1 Comparison of LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons reach up to an octahedral morphology. Compared with the undoped LiMn<sub>2</sub>O<sub>4</sub> anoparticles, the Si-doped LiMn<sub>2</sub>O<sub>4</sub> sample can show higher initial discharge capacity, and

the octahedral morphology can enhance the cycling stability of  $LiMn_2O_4$ . The Si-doping can provide the more expanded and regular  $MnO_6$  octahedra, which achieves the better accommodation of Mn(III)–Mn(IV) interconversion during the electrochemical cycling process [17, 20]. Moreover, the octahedral morphology enhances the structure stability by suppressing the manganese dissolution [22, 26, 27]. The above positive function contributes to the good electrochemical performance of  $LiSi_{0.05}Mn_{1.95}O_4$  octahedrons.

**Table 1** Comparison of LiSi0.05Mn1.95O4 octahedrons and LiMn2O4-based cathode materials reported by other works [17, 20, 22].

Sample	Cycling condition	Initial capacity (mAh g <sup>-1</sup> )	Capacity retention (%)	Reference
LiMn <sub>2</sub> O <sub>4</sub> nanoparticles	1.0 C, 100 cycles	114.0	87.8	[22]
Si-doped LiMn <sub>2</sub> O <sub>4</sub> particles	0.5 C, 100 cycles	134.6	85.1	[20]
LiSi <sub>0.05</sub> Mn <sub>1.95</sub> O <sub>4</sub> particles	1.0 C, 300 cycles	139	75	[17]
LiSi <sub>0.05</sub> Mn <sub>1.95</sub> O <sub>4</sub> octahedrons	1.0 C, 100 cycles	118.7	94.7%	This work



Figure 7. Rate stability of the undoped LiMn<sub>2</sub>O<sub>4</sub> particles and LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons.

Figure 7 presents the rate performance of the undoped  $LiMn_2O_4$  particles and  $LiSi_{0.05}Mn_{1.95}O_4$  octahedrons. It can be found that the discharge capacity is intimately connected to the cycling rate. When cycled at the low rate of 0.5 C, the discharge capacities of the undoped  $LiMn_2O_4$  particles and  $LiSi_{0.05}Mn_{1.95}O_4$  octahedrons have no obvious difference.



**Figure 8**. Cycling performance of the undoped LiMn<sub>2</sub>O<sub>4</sub> particles and LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons at 55 °C.



Figure 9. Rate capability of the undoped LiMn<sub>2</sub>O<sub>4</sub> particles and LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons at 55 °C.

However, as the cycling rate increases, the corresponding discharge capacity gradually show obvious change. When the cycling rate increases to 5.0 C, the undoped LiMn<sub>2</sub>O<sub>4</sub> particles exhibit fairly small discharge capacity of 52.3 mAh g<sup>-1</sup>, which is 45.2% of the initial discharge capacity of the undoped LiMn<sub>2</sub>O<sub>4</sub> at 0.5 C. For the LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons, the corresponding discharge capacity can reach up to 92.5 mAh g<sup>-1</sup>, which is 75.1% of the initial discharge capacity of this Si-doped spinel at 0.5 C. Such good performance has much to do with the synergistic effect of Si-doping and octahedral morphology of the LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons [17, 23, 27].

Figure 8 presents the cycling performance of the undoped LiMn<sub>2</sub>O<sub>4</sub> particles and LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons at 55 °C, and the cycling rate is set to be 1.0 C. It can be found that the combination of Sidoping and octahedral morphology performs an important role in the improvement of high-temperature electrochemical performance. The undoped LiMn<sub>2</sub>O<sub>4</sub> particles exhibit poor cycling stability with lower capacity retention rate of 58.9%. For the LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons, the cycling stability was significantly improved with higher discharge capacity of 113.5 mAh g<sup>-1</sup>, and the corresponding capacity retention rate reaches up to 95.3%. Figure 9 shows the high-temperature rate performance of the undoped LiMn<sub>2</sub>O<sub>4</sub> particles and LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons at 55 °C. It can be found that the discharge capacity is intimately connected to the cycling rate. When cycled at the low rate of 0.5 C, both LiMn<sub>2</sub>O<sub>4</sub> particles and LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons present similar discharge capacities (115.9 mAh g<sup>-1</sup> and 123.4 mAh g<sup>-1</sup>). When the cycling rate increases to 5.0 C, the discharge capacity of the undoped LiMn<sub>2</sub>O<sub>4</sub> particles rapidly decreases to 56.9 mAh g<sup>-1</sup>. For the LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons, the corresponding discharge capacity can reach up to 98.2 mAh g<sup>-1</sup>. The above result further confirms the positive function of Sidoping and octahedral morphology on the electrochemical performance LiMn<sub>2</sub>O<sub>4</sub>.

## 4. CONCLUSIONS

The LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons were prepared by high-temperature solid-state method with Mn<sub>3</sub>O<sub>4</sub> octahedrons as manganese precursor and SiO<sub>2</sub> nanoparticles as silicon dopant. The comodification strategy of Si-doping and octahedral morphology effectively enhances the electrochemical performance of LiMn<sub>2</sub>O<sub>4</sub>. The research result indicated that the Si-doping could provide more expanded and regular MnO<sub>6</sub> octahedrons, which achieved the good accommodation of Mn(III)–Mn(IV) interconversion during the electrochemical cycling process. Moreover, the octahedral morphology enhanced the structure stability by suppressing the manganese dissolution. The LiSi<sub>0.05</sub>Mn<sub>1.95</sub>O<sub>4</sub> octahedrons exhibited good cycling stability and rate performance, which may promote the widespread use of LiMn<sub>2</sub>O<sub>4</sub>.

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