Electrochemical Corrosion and Electrical Resistivity of Reinforced Recycled Aggregate Concrete

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It is planned to improve the durability of reinforced concrete structures and the total or partial replacement of their components by the use of recyclable materials; as strategies to contribute to the concrete industry sustainability. The electrical resistivity and its relationship with the microstructure and electrochemical corrosion resistance in reinforced concrete manufactured with sustainable characteristics, that is, with recycled aggregates and supplementary cementitious materials; was evaluated in this study. The electrical resistivity in reinforced concretes is closely related to the microstructure of the cementitious matrix, and with the pore structure and distribution. As a consequence of the their components nature, it is assumed that the proposed sustainable concretes present different microstructure and porosity than the conventional concretes; because of this, the electrical and electrochemical response of those systems was analyzed with the help of the electrochemical impedance spectroscopy technique.

Keywords: Recycled aggregate concrete, Corrosion, Concrete resistivity, Poozolanic materials.

1. INTRODUCTION

Nowadays, as a result of the significant environmental impact caused by the manufacture process of concretes and, as contribution to this industry, sustainability, it is being implementing improvements in its durability, as well as, the replacement of their components by less-aggressive recyclable materials. Some of the alternative materials more used in the concrete manufacture are those

which partially replace the Portland cement (Supplementary Cementitious Material [SCM]), i.e. silica fume (SF), fly ash (FA) and the blast furnace slag; concerning the aggregates, the recycled concrete aggregates are employed as replacement of the natural aggregates. At present there are an important number of works about the behavior and mechanical performance of concretes manufactured with recycled aggregates [1-10]; on the other hand and more recently, concretes with addition of SCM are currently providing knowledge [11-15], giving emphasis to the favorable impact of SCM in the mechanical behavior and in the environmental benefit. Due to their components nature, the concrete manufactured with recycled material present different microstructure and greater porosity than the conventional concrete [16-19]; it is also known that the electrical resistivity of the concrete is related to the microstructure of the cementitious matrix and to their pores distribution [20-22]. This last property has been recognized as an important parameter to evaluate the concrete durability [23].

In order to verify the behavior of reinforced concrete manufactured with recycled aggregates and with SCM, the electrical and electrochemical response are analyzed by means of the Electrochemical Impedance Spectroscopy technique (EIS), recognized as an efficient method to evaluate the dielectric and electrochemical properties in resistive systems (as the case of reinforced concrete). The basics principles of electrochemical corrosion of reinforced concrete is well known [24]

2. EXPERIMENTAL PART

It was produced four sets of specimens with water-cementitious material ratio of 0.48: reference series made with natural aggregate (NA) and 100% Compound Portland Cement (CPC) [25], series made with recycled coarse aggregate (RCA) and 100 % CPC, series made with RCA and 30% FA as a partial replacement of the CPC, and finally, series made with RCA and 10% SF as a partial replacement of the CPC.

2.1 Materials

RCA comes from the crushing of concrete specimens made with natural aggregates, CPC with a water-cement ratio of 0.50 and curing for a 28-day period with controlled temperature and relative humidity (RH) of 23 ± 2 °C and 98 ± 1 %. Natural aggregates come from quarry crushed rock (coarse) and river sand (fine). Table 1 shows some of the physical properties of these aggregates.

Table 1. Pphysical properties of the aggregates used in concrete mixtures

Type of aggregate	Relative density (g/cm ³)	Absorption (%)	Humidity (%)	Fineness module (%)	Maximum size (mm)
Recycled coarse	2.19	6.55	2.14	-	19
Natural coarse	2.50	0.44	0.28	-	19
Natural fine	2.43	4.08	6.66	2.73	4.76

As SCM, Mexican Class F FA according to ASTM C618 [26] and SF North-American with the requirements of ASTM C1240 [27] were used. Table 2 shows the physical and chemical properties of these materials.

Chemical composition (% of weight)								
Material	SiO ₂	Al_2O_3	Fe ₂ O ₃	CaO	SO ₃	K ₂ O	Na ₂ O	MgO
CPC	19.94	4.40	2.97	63.50	3.08	0.42	0.12	-
FA	58.84	16.72	3.52	7.35	0.13	0.79	0.94	1.76
SF	95.22	0.08	2.37	0.26	0.11	0.56	0.30	0.24
Physical properties								
	Density (g/cm ³) Specific		Specific s	surface, BET (m ² /kg)		Average size (µm)		
CPC	3.15		1400		15-25			
FA	2.	2.35		1200		5-15		
SF	2.	2.27		19600		0.1-0.2		

Table 2. Physico-chemical properties of cementitious materials.

2.2 Concrete Resistivity and Electrochemical Corrosion

To evaluate the electrical resistivity and corrosion resistance of the systems studied were prepared two cylindrical specimens (length=30 cm, diameter = 15 cm) for each mixture, its characteristics and proportions are presented in Table 3. These specimens were cured for 28 days in a chamber with temperature of 23 ± 2 °C and $98 \pm 1\%$ of RH.

Table 3. Characteristics and proportion of the test mixtures (by $1m^3$ of concrete).

Materials	Mixtures identification						
(Kg)	Natural coarse	Recycled coarse aggregate, natural sand and SCM					
	and fine						
	aggregate						
	NA 100% CPC	RA 100% CPC	RA 30% FA	RA 10% SF			
Water	213.31	213.31	213.31	213.31			
Gravel	994.55	870.58	870.58	870.58			
Sand	766.17	915.35	915.35	915.35			
Cement	444.44	444.44	311.11	400.00			
SCM	0.000	0.000	133.33	44.44			

Each specimen had two embedded bars of 1018 carbon steel $\Phi = 9.5$ mm and with an exposed area of 64 cm², localized 4.5 cm from the outside. The steel bars were cleaned before placement for corrosion test. The specimens were exposed to an aqueous solution of 3.5% NaCl to assess both: the

change in the electrolytic resistance (Re), which has correlation with the concrete resistivity (ρ); and the variation in the Resistance of charge transfer (Rct), which has correlation with the corrosion current density (i_{corr}) of reinforcement.

The evaluation of parameters Re and Rct, was accomplished by EIS in a potentiostat/Galvanostat/FRA of ACM Instruments, the test parameters were a 10 mV amplitude potential to maintain the system linearity in a frequency range of 1 mHz to 10 kHz. The experimental testing is shown in Figure 1.

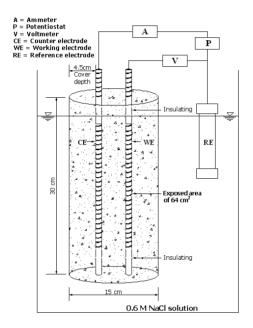


Figure 1. Experimental scheme for testing EIS.

The EIS results were represented with Nyquist diagrams [Figure 2 a)], and an equivalent circuit (EC) Randles type (shown in Figure 2 b) was used to determine accurately the intersection value at low frequencies and can calculate the Rct.

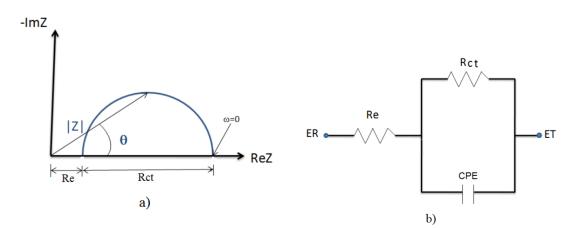


Figure 2. a) Ideal Nyquist diagram, b) EC used to simulate experimental results of EIS.

The calculated Rct and the Re obtained from the intersection at high frequencies with the Nyquist diagram were used to determine i_{corr} and concrete electrical resistivity (ρ), respectively. Using Equation 1 (27), i_{corr} , was calculated; where B is the Tafel constant with recommended value [29-32] of 0.052 V for the passive corrosion of steel in concrete; ρ was determined from Equation 2, where Cc is a cell constant that depends on the geometry of the conducting body [33].

$$i_{corr} = \frac{B}{Rct} \tag{1}$$

$$\operatorname{Re} = \rho \cdot C_c \tag{2}$$

 C_c is a cell constant (58,47 cm), L is the submerged length of the reinforced concrete specimen (20 cm), D is the concrete specimen diameter (15 cm), d is the steel reinforcement diameter (0.95 cm) and z is the distance between the concrete specimen and reinforcement (5.05 cm). The cell constant C_c was determined analytically (Equation 3) applying the principles of heat conduction through the elements proposed by Sunderland and Johnson [34]. The Laplace equation was resolved considering as limit the steel elements temperature (Reinforcement) T_1 , and the concrete elements temperature T_2 , where $T_1 = T_2 = \text{constant}$.

$$C_{c} = \left(\frac{2\pi L}{\cosh^{-1}(\frac{D^{2} + d^{2} - 4z^{2}}{2Dd})}\right)$$
(3)

3. RESULTS AND DISCUSSION

Dielectric and electrochemical properties of concrete-steel systems with a 90-days exposure are represented in the Nyquist diagram of Figure 3; it shows that the RA 100% CPC system has minimum values of Rct and Re, while the RA 10% SF system reaches the maximum. The ρ as a function of the exposure time, shown in Figure 4, denotes that the use of RCA results from a decrement in the concrete resistivity. In both systems, NA 100% CPC and RA 100% CPC, ρ unsurpassed the limit of 10 k Ω -cm indicating that it is not the controlling parameter of corrosion rate; and therefore there is a high risk that the phenomenon should present in them [22,35-37]. In the SCM systems, it is remarkable the positive effect of them on the concrete resistivity; the ρ in the RA 30% FA system is slightly higher than the limit value, while the RA 10% SF system exceeds it largely, allowing to predict a moderate or low likelihood of beginning the corrosion. The increase in ρ in SCM systems is attributed to the densification of porous system (because the ionic conduction is through pore structure of cementitious matrix), generated by the increase in the formation of calcium silicate hydrates (CSH) and the reduction in the formation of Ca(OH)₂ product of the pozzolanic reaction between the last one with the SiO₂ content in the SCM. Table 2 shows that the SF is approximately 100 times finer than cement and

FA, providing much SiO₂. Because hydration and pozzolanic reactions are surface phenomena, it is deduced that the microstructure of concrete with SF becomes denser than the rest of the systems due to the production of more CSH which replaces CH and generates pore refinement, and therefore, it reports an increase in its electrical resistivity. Moreover, in the first three months of the test, the ρ in SCM systems increases rapidly, whereas after this period it slows down, thus suggesting that such reactions come to equilibrium after about a 90-day period of exposure.

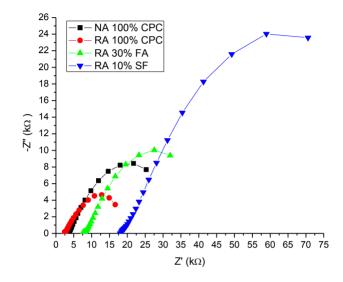


Figure 3. Nyquist diagram for concrete-steel systems within 90 days of exposure.

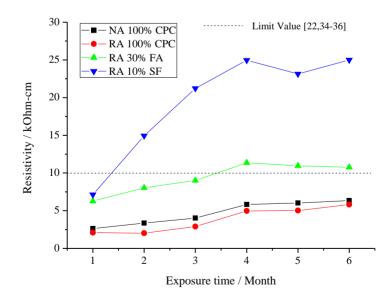


Figure 4. Evolution of electrical resistivity as a function of exposure time

The determined results of icorr, from the obtained values of Rct, are shown in Figure 5, in this figure is possible to compare the corrosion resistance values in concrete-steel systems used in this study. In a general perspective, during the third-test month, an inflexion point is located between two corrosion states with different activity level; this division is attributed to, from this point, the reactions of hydration-pozzolanic in the cement and SCM reach equilibrium and their speed is stabilized. It is also noted that the system with greater corrosion resistance is the RA 10% SF, since despite its corrosive activity was greatest in the first month, it decreased significantly during the subsequent months until reaching a minimum level of corrosion (0.004 μ A/cm²) at the sixth month of trial, with an average of 0.025 uA/cm² in all its trial period. Moreover, the RA 100% CPC system reported its maximum corrosion levels (from 0.044 to 0.060 μ A/cm²) until the third month, although in the last three months its corrosive activity is significantly reduced in an average of i_{corr} for this period of 0.031 μ A/cm², classifying it as the system with less corrosion resistance. The RA 30% FA system reported low corrosion levels till the second test month (from 0.022 to 0.037 μ A/cm²), maintaining this activity at the same level for the subsequent months, the average i_{corr} was 0.025 μ A/cm², similar to that obtained by the RA 10% SF system. The mentioned results show that although there is a significant difference in terms of ρ in the SCM systems, the corrosion resistance can be equivalent. This is attributable to the high content of Al₂O₃ (\cong 17%) in the FA with respect to SF (\cong 0.1%) and cement (≅4%). In a X-rays study by Koulombi et al [38], it was demonstrated that the addition of FA generates a higher content of chloroaluminate (Friedel salt), and consequently, lower levels of free chlorides, which are responsible in this case of the localized reinforcing steel corrosion.

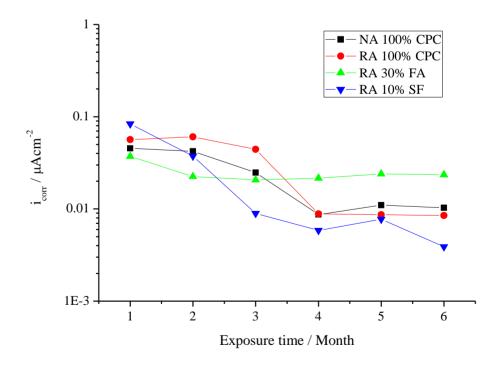


Figure 5. Variation in the corrosion current density as a function of the exposure time.

4. CONCLUSIONS

- The use of 100% recycled aggregate, decreases concrete electrical resistivity, increasing then the corrosion rate of reinforcement.
- Fly ash and silica fume contribute significantly in increase the concrete electrical resistivity, due to cementing matrix densification and pore refinement.
- Although the magnitude of the electrical resistivity in a concrete with silica fume, is two times larger than that of a concrete with fly ash; the corrosion resistance in both is very similar due to the high content of Al_2O_3 in the fly ash compared to silica fume.
- The use of fly ash and silica fume increase the durability of concretes with 100% recycled coarse aggregate, being this feasible to use in structures exposed to aggressive environments, contributing to the concrete industry sustainability.

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References

- 1. A. Ajdukiewicz and A. Kliszczewicz, Cement and Concrete Composites, 24 (2002) 269.
- 2. H. Chen, T. Yen and K. Chen, *Cement and Concrete Research*, 33 (2003) 125.
- 3. A. Katz, Cement and Concrete Research, 33 (2003) 703.
- 4. C.S. Poon, Z.H. Shui, L. Lam, H. Fok and S.C. Kou, *Cement and Concrete Research*, 34 (2004) 31.
- 5. I. Topçu and S. Sengel, Cement and Concrete Research, 34 (2004) 1307.
- 6. T. Tu, Y. Chen and C. Hwang, *Cement and Concrete Research*, 36 (2006) 943.
- 7. I. Martínez and C. Mendoza, Ingeniería Investigación y Tecnología, 7 (2006) 151.
- 8. K. Rahal, Building and Environment, 42 (2007) 407.
- 9. M. Casuccio, M.C. Torrijos, G. Giaccio and R. Zerbino, *Construction and Buildings Materials*, 22 (2008) 1500.
- 10. A.K. Padmini, K. Ramamurthy and M.S. Mathews, *Construction and Buildings Materials*, 23 (2009) 829.
- 11. K.Y. Ann, H.Y. Moon, Y.B. Kim and J. Ryou, Waste Management, 28 (2008) 993.
- 12. B. González and F. Martínez, Building and Environment, 43 (2008) 429.
- 13. S.C. Kou, C.S. Poon and D. Chan, Materials and Structures, 41 (2008) 1191.
- 14. M.L. Berndt, Construction and Buildings Materials, 23 (2009) 2606.
- 15. V. Corinaldesi and G. Moriconi, Construction and Buildings Materials, 23 (2009) 2869.
- 16. J.M.V. Gómez, Cement and Concrete Research, 32 (2002) 1301.
- 17. C.S. Poon, Z.H. Shui and L. Lam, Construction and Buildings Materials, 18 (2004) 461.
- 18. V.W.Y. Tam, X.F. Gao and C.M. Tam, Cement and Concrete Research, 35 (2005) 1195.
- 19. M. Etxeberría, E. Vazquez and A. Mari, Magazine of Concrete Research, 58 (2006) 683.
- 20. P.J. Tumidajski, A.S. Shumacher, S. Perron, P. Gu and J.J. Beaudoin, *Cement and Concrete*. *Research*, 26 (1996) 539.
- 21. P.J. Tumidajski, Cement and Concrete Research, 35 (2005) 1262.
- 22. R. Polder, C. Andrade, B. Elsener, O.E. Vennesland, J. Gulikers, R. Weidert and M. Raupach, *Materials and Structures*, 33 (2000) 603.

- 23. Song H.W. and Saraswathy, International Journal of Electrochemical Science, 2 (2007) 1.
- 24. Dao L.T.N., Dao V.T.N., Kim S.H. and Ann K.Y., *International Journal of Electrochemical Science*, 5 (2010) 302.
- 25. ONNCCE, NMX-C-414-04, *Especificaciones y métodos de prueba del cemento hidráulico*, Instituto Mexicano del Cemento y el Concreto México, (2004).
- 26. ASTM International, ASTM C618-99 (1999).
- 27. ASTM International, ASTM C1240-05 (2005).
- 28. M. Stern and A. Geary, Journal of the electrochemical society, 104 (1957) 56.
- 29. R.K. Dhir, M.R. Jones and M.J. McCarthy, Cement and Concrete Research, 23 (1993) 1443.
- J.A. González, E. Ramírez, A. Bautista and S. Feliú, *Cement and Concrete Research*, 26, (1996) 501.
- 31. K.R. Gowers and S.G. Millard, Corrosion Science, 35 (1993) 1593.
- 32. P.S. Mangat and B.T. Molloy, Materials and Structures, 25 (1992) 404.
- 33. J.M. Torrents, J. Roncero and R. Gettu, Cement and Concrete Research, 28 (1998) 1325.
- 34. J.E. Sunderland and K.R. Johnson, ASHRAE 71st Annual Meeting, Cleveland (1964).
- 35. C. Andrade and C. Alonso, Construction and Building Materials, 10 (1996) 15.
- 36. R.D.Browne, Durability of Building Materials, 1 (1982) 113.
- 37. P.G.Cavalier and P.R.Vassie, Proceedings of Institution of Civil Engineers, 70 (1981) 461.
- 38. N. Koulombi, G. Batis and C.H. Malami, *Progress in the understanding and prevention of corrosion, Mercer AD editors, Institute of Materials, Spain,* (1993) 619

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